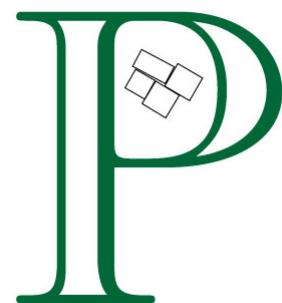


# Rietveld Profile Functions



**NORTH CENTRAL  
COLLEGE 1861**

James A. Kaduk  
Illinois Institute of Technology  
North Central College  
[Kaduk@polycrystallography.com](mailto:Kaduk@polycrystallography.com)

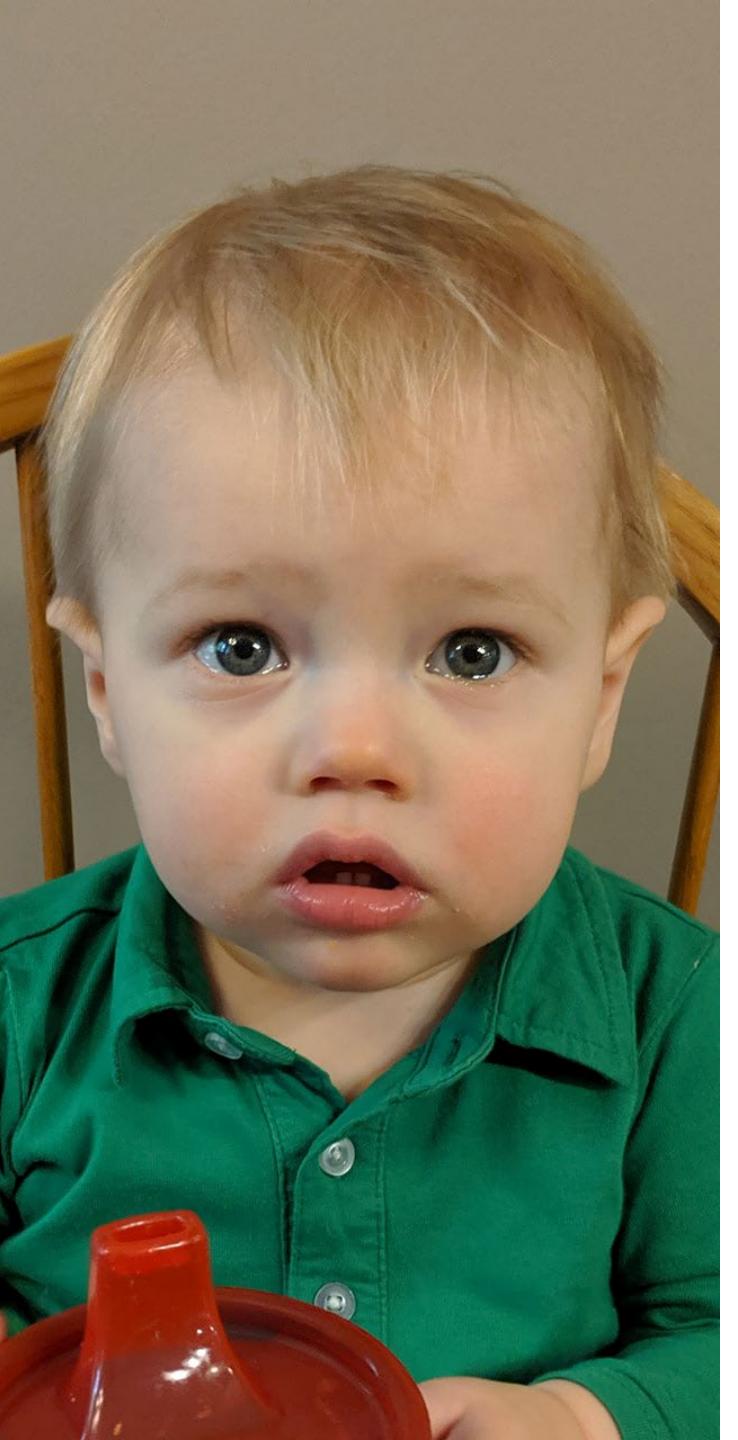


ILLINOIS INSTITUTE  
OF TECHNOLOGY



# Mottoes for the powder diffractionist

- It depends
- Overlap kills
- *Everything's* a sample
- Desperate analysts do desperate things
- If you have a single crystal, you should use it



# Convolution

$$(f * g)(t) = \int_{-\infty}^{\infty} f(\tau) \cdot g(t - \tau) d\tau = \int_{-\infty}^{\infty} f(t - \tau) \cdot g(\tau) d\tau$$

The Fundamental Parameters approach

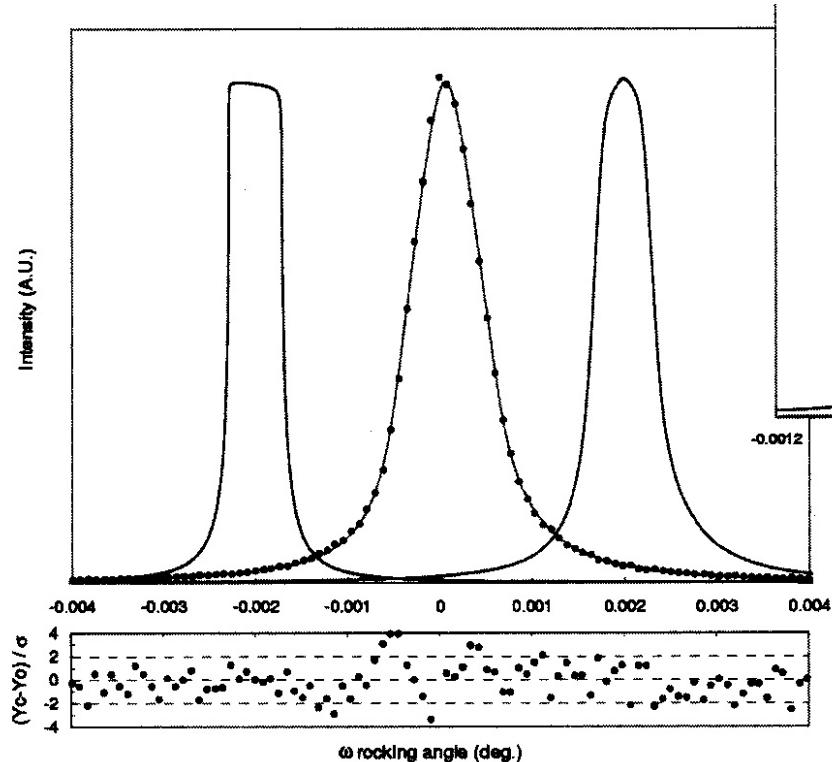
<http://en.wikipedia.org/wiki/Convolution>

# Convolution

- Convolution of one function (input) with a second function (impulse response) gives the output of a system
- A weighted moving average
- In optics, “blur” is described by convolution

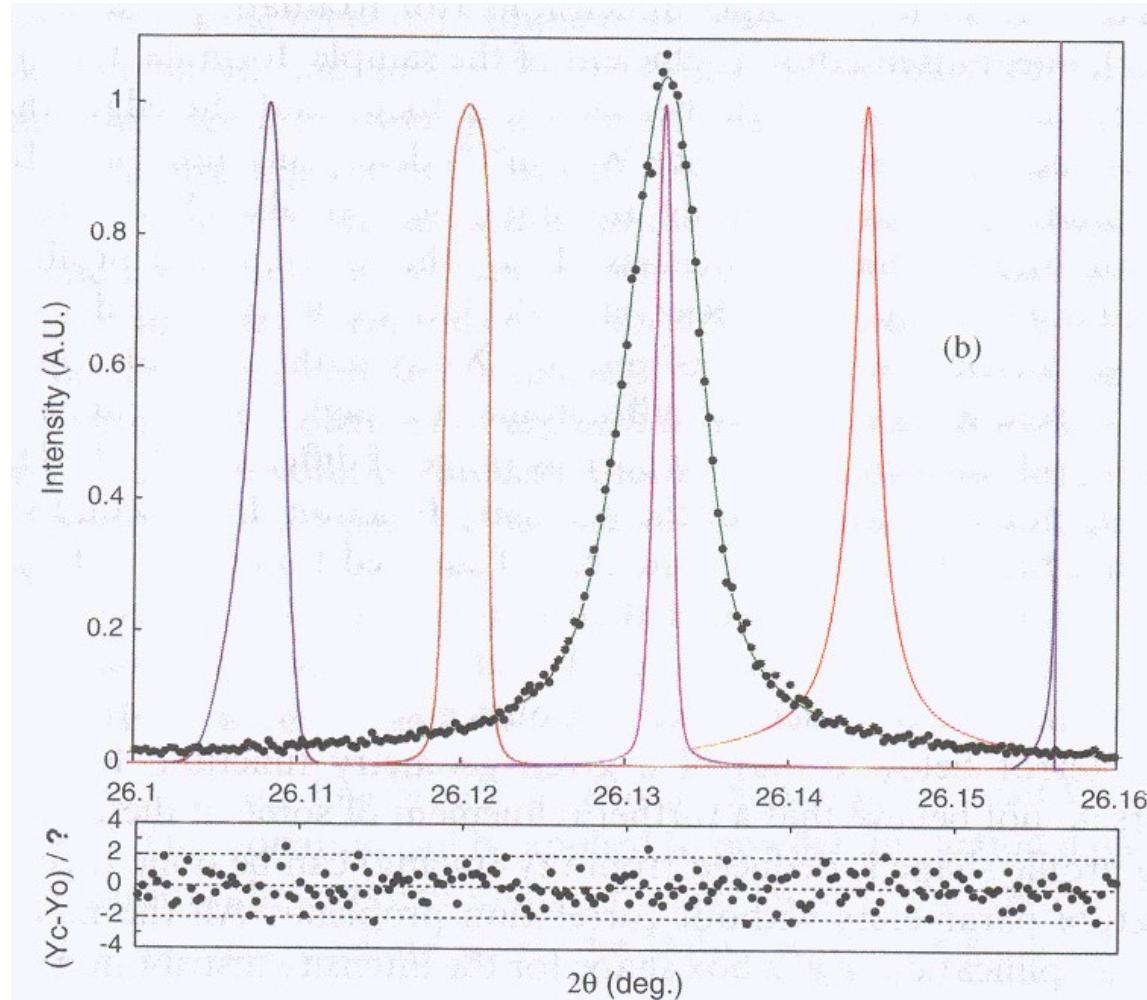
BHT working on convolution to generate instrument parameters

# ESRF BM16 (now ID22) Second Monochromator Crystal Rocking Curve



Left; perfect Si(111) Darwin profile. Right: perfect Si(111) reflection convoluted with first crystal strain function. Center: experimental data and fit by convolution of left and right curves.

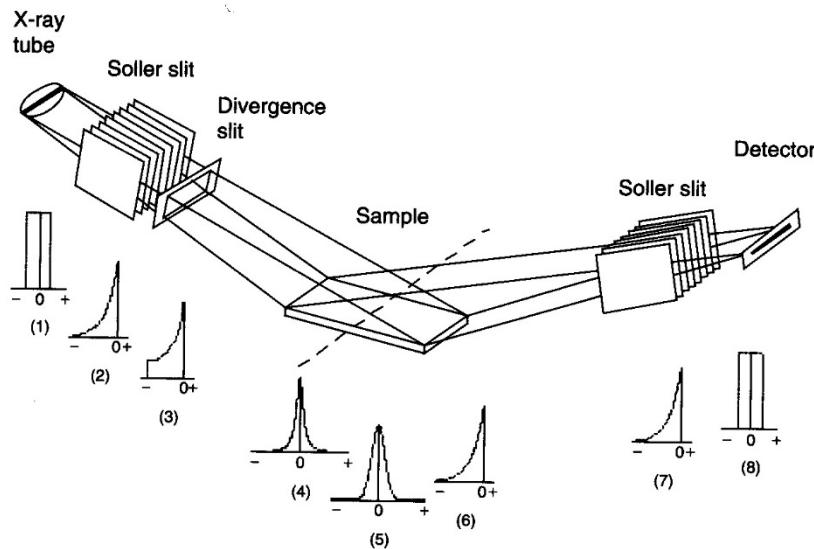
O. Masson, E. Dooryhee, and A. N. Fitch, “Instrument line-profile synthesis in high-resolution synchrotron powder diffraction”, J. Appl. Cryst., 36, 286-294 (2003).



$\text{Na}_2\text{Ca}_3\text{Al}_2\text{F}_{14}$  (921) reflection. From left to right: the incident beam source profile, the transfer function of the monochromator, the pure sample profile, the reflection profile of the analyzer, and the axial divergence asymmetry function.

Masson, Dooryhee, and Fitch, in A. Le Bail, “The Profile of a Bragg Reflection for Extracting Intensities”, in R. E. Dinnebier and S. J. L. Billinge, *Powder Diffraction: Theory and Practice*, RSC Publishing (2008).

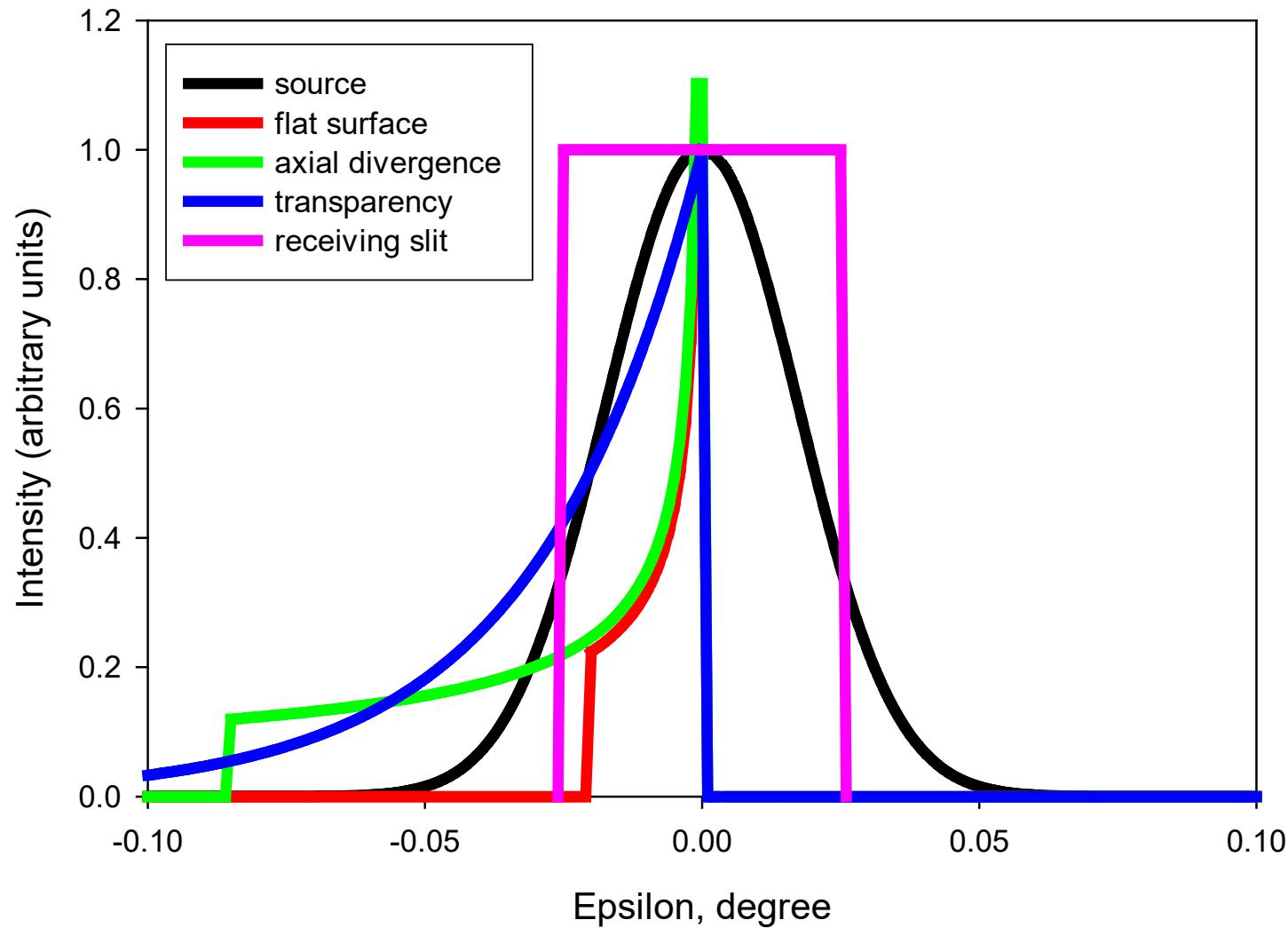
# Bragg-Brentano Diffractometer



**Figure 4.25** Schematic representation of the fundamental parameters approach for a divergent beam diffractometer showing the principal optical components and the sample together with their related aberration functions as discussed in Section 2.2.2 including (1) finite X-ray source width, (2) primary axial divergence, (3) horizontal divergence, (4) crystallite size, (5) strain, (6) absorption, (7) secondary axial divergence and, (8) receiving slit width. Figure copyright Bruker AXS.

A. Kern, “Profile Analysis”, in A. Clearfield, J. Reibenspies, and N. Bhuvanesh, *Principles and Applications of Powder Diffraction*, Wiley (2008).

# Profile Contributions



# Profile Contributions

Effect	Equation	Range
X-ray Source	$\exp(-k_1^2 \varepsilon^2)$ $k_1 = 1.67(\text{FWHM})$	$-\infty$ to $+\infty$
Flat Surface	$ \varepsilon ^{-1/2}$	$-(\gamma^2 \cot \theta)/114.6$ to 0 $\gamma$ = divergence
Axial Divergence	$ 2\varepsilon \cot \theta ^{-1/2}$	$-(\delta^2 \cot \theta)/(4 \times 57.3)$ to 0 $\delta$ = axial divergence
Transparency	$\exp(k_4 \varepsilon)$ $k_4 = (4\mu R/114.6)\sin 2\theta$	$-\infty$ to 0
Receiving Slit		$-(\text{FWHM})/2$ to $+(\text{FWHM})/2$

H. P. Klug and L. E. Alexander, *X-ray Diffraction Procedures*, Wiley (1974).

# Plus the Cu K $\alpha$ Profile

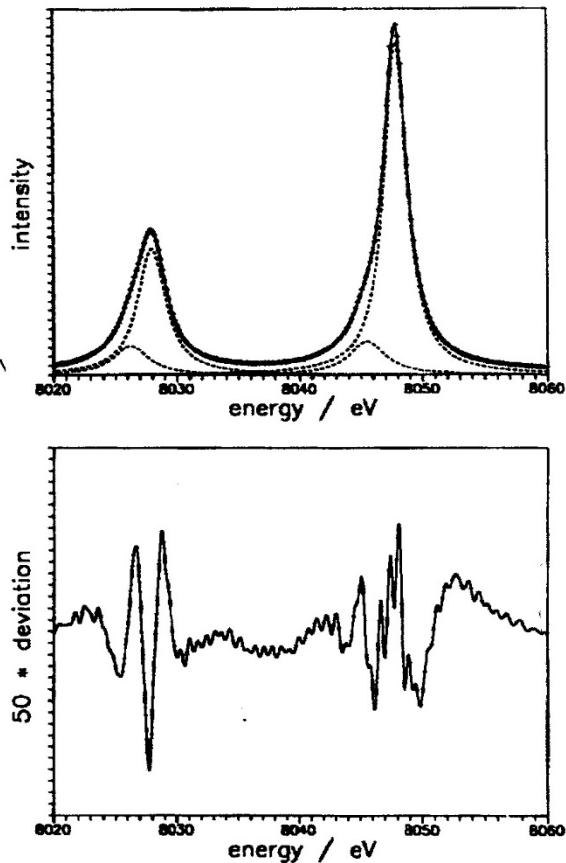


Fig. 4. Top: Cu K $\alpha$  spectrum measured at the Si 444 reflection with the single-crystal spectrometer (crosses) and fitted curve (solid) consisting of four Lorentzians (dotted); bottom: absolute deviation of the fitted curve (50 times enlarged).

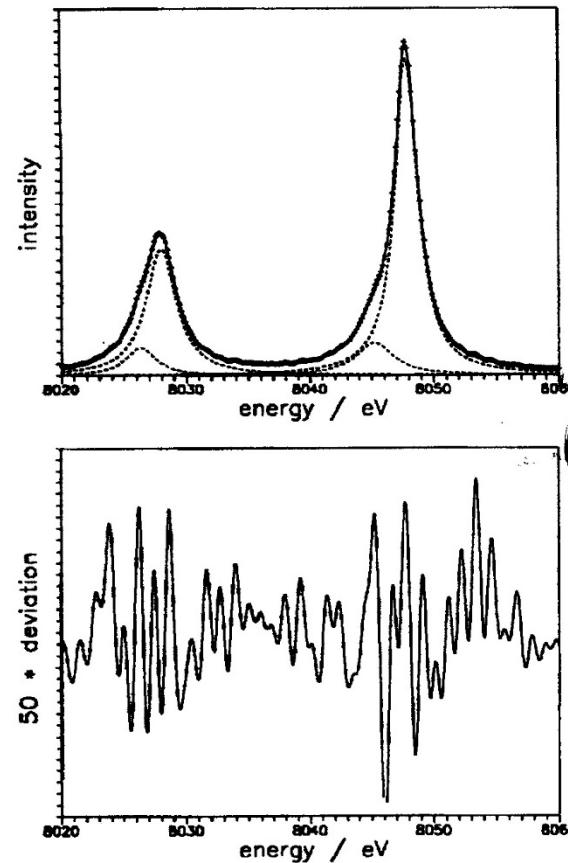
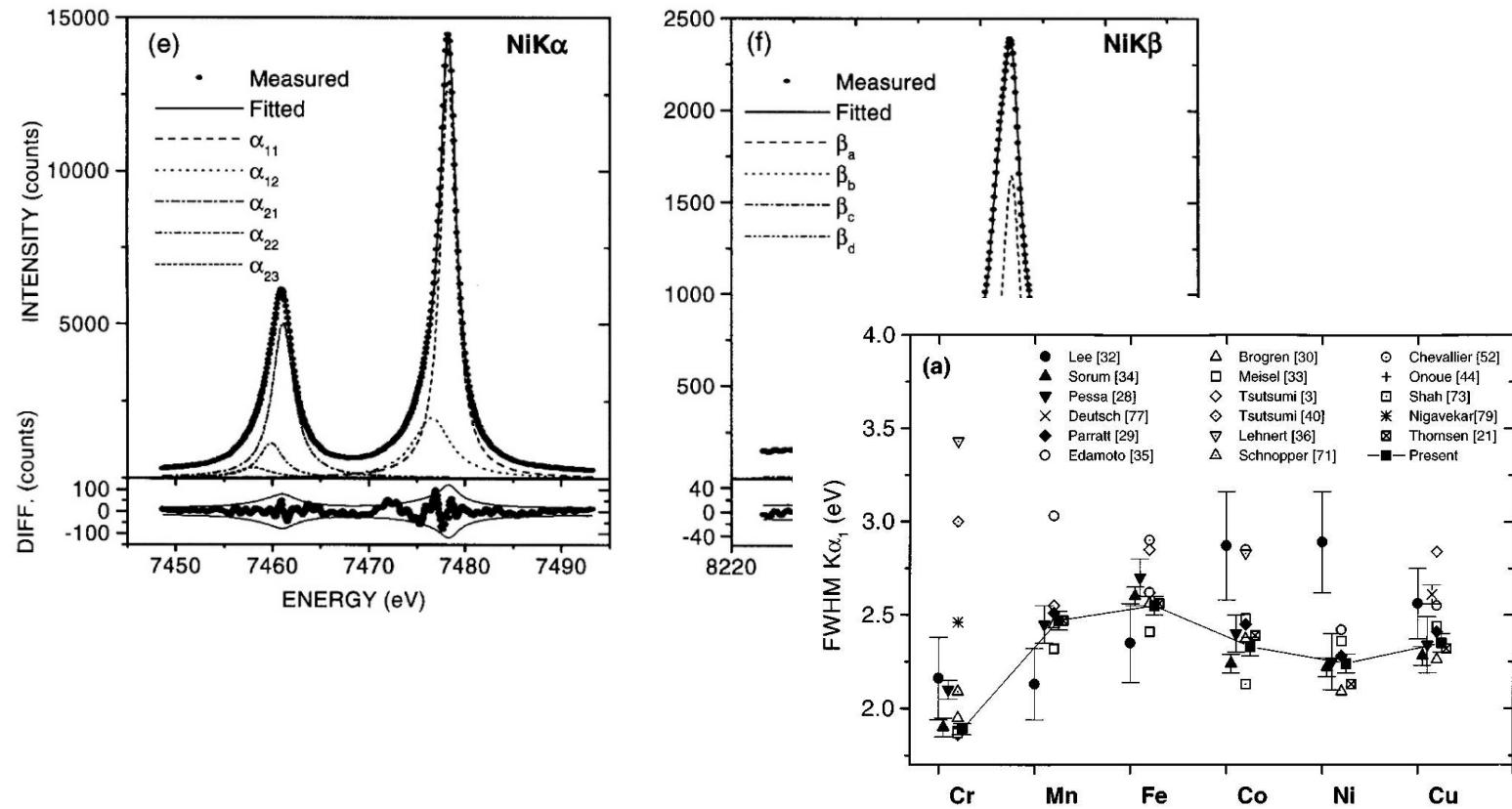


Fig. 5. Top: Cu K $\alpha$  spectrum measured at the Si 333 reflection with the double-crystal spectrometer (crosses) and fitted curve (solid) consisting of four Lorentzians (dotted); bottom: absolute deviation of the fitted curve (50 times enlarged).

J. Hartwig, G. Hölzer, J. Wolf, and E. Förster, "Remeasurement of the Profile of the Characteristic Cu K $\alpha$  Emission Line with High Precision and Accuracy", *J. Appl. Cryst.*, **26**, 539-548 (1993).

# More on Emission Profiles



G. Hölzer, M. Fritsch, M. Deutsch, J. Härtwig, and E. Förster, “ $\text{K}_{\alpha 1,2}$  and  $\text{K}_{\beta 1,3}$  emission lines of 3d transition metals”, *Phys. Rev. A*, **56**, 4554-4568 (1997).

Multiconfiguration Dirac-Fock calculations in open-shell atoms: Convergence methods and satellite spectra of the copper K $\alpha$  photoemission spectrum,  
C. T. Chantler, J. A. Lowe, and I. P. Grant, Phys. Rev. A, 82, 052505 (2010).

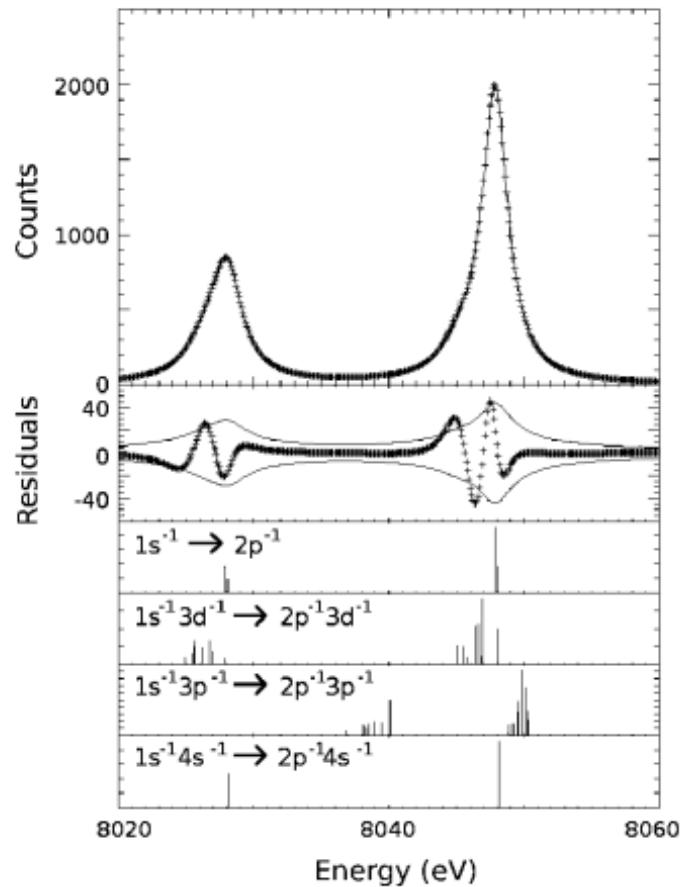
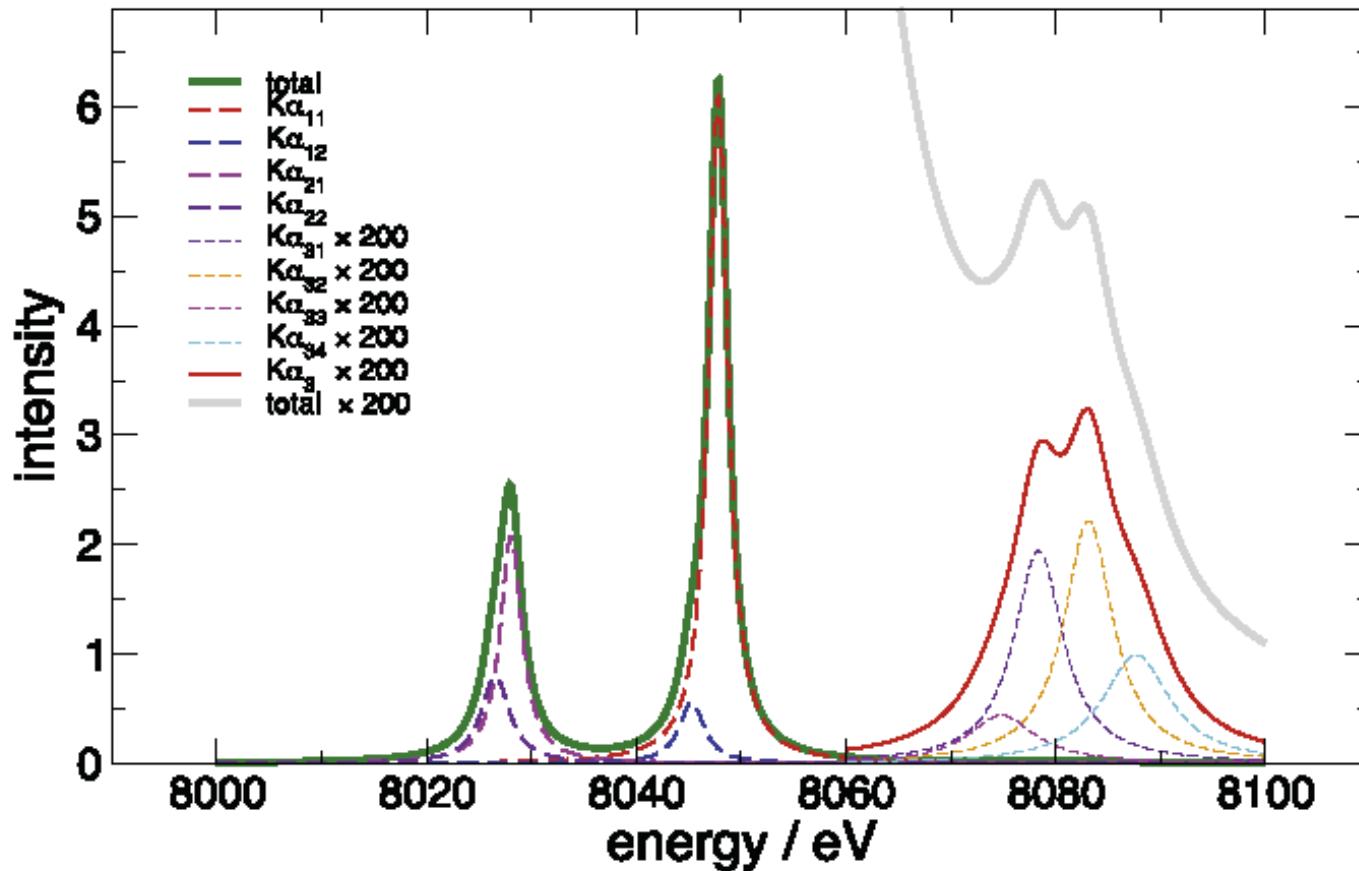


FIG. 1. Experimental and fitted theoretical spectrum for copper K $\alpha$ . The curve bounding the residuals is  $\pm\sigma$ . The positions of the stick diagrams represent the transition energies contributing to the spectrum, and the height represents the intensity, normalized to the most intense transition of the group. The energy of the 4s spectator transition clearly indicates why approximate treatment of the open shell has provided good results in previous work.

High-precision measurement of the x-ray Cu K $\alpha$  spectrum, M. H. Mendenhall, A. Henins, L. T. Hudson, C. I. Szabo, D. Windover, and J. P. Cline, *J. Phys. B: At. Mol. Opt. Phys.* **50**, 115004 (2017)



**Figure 19.** Separated peak components from the fit.

Hugo Rietveld's low-resolution neutron diffraction peaks were Gaussian (determined mainly by the neutron spectral distribution, the monochromator response function, and the divergences of the Soller collimators).

He used the “Caglioti” function to describe the widths.

H. M. Rietveld, “A Profile Refinement Method for Nuclear and Magnetic Structures”, *J. Appl. Cryst.*, **2**, 65-71 (1969).

G. Caglioti, A. Paoletti, and F. P. Ricci, “Choice of Collimators for a Crystal Spectrometer for Neutron Diffraction:, *Nucl. Inst.*, **3**, 223-228 (1958).

$$\text{FWHM}^2 = U \tan^2 \theta + V \tan \theta + W$$

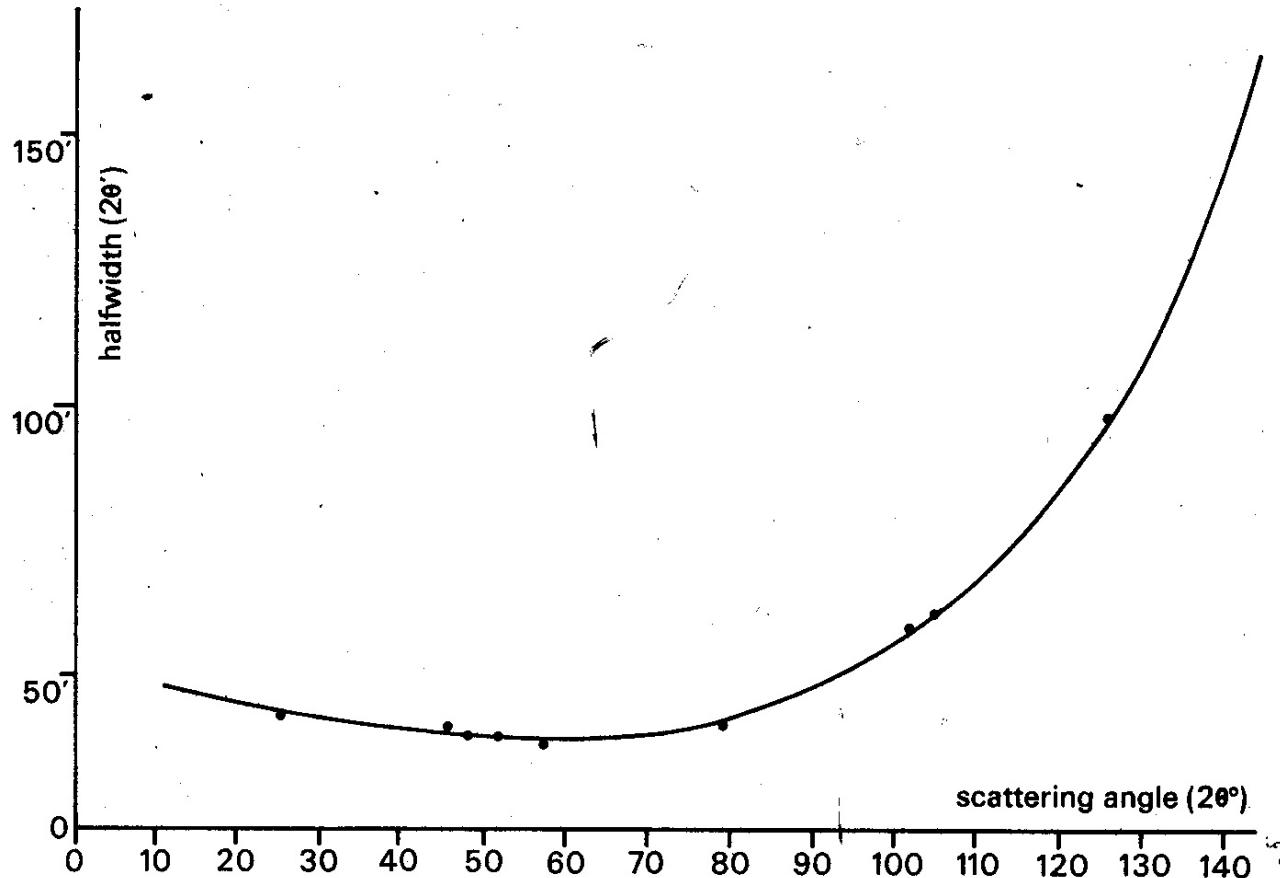


Fig. 3. Variation of peak width with Bragg angle; .... measured halfwidths, — calculated curve.

$$\text{FWHM}^2 = A \tan^2 \theta + B \tan \theta + C + D \cot^2 \theta$$

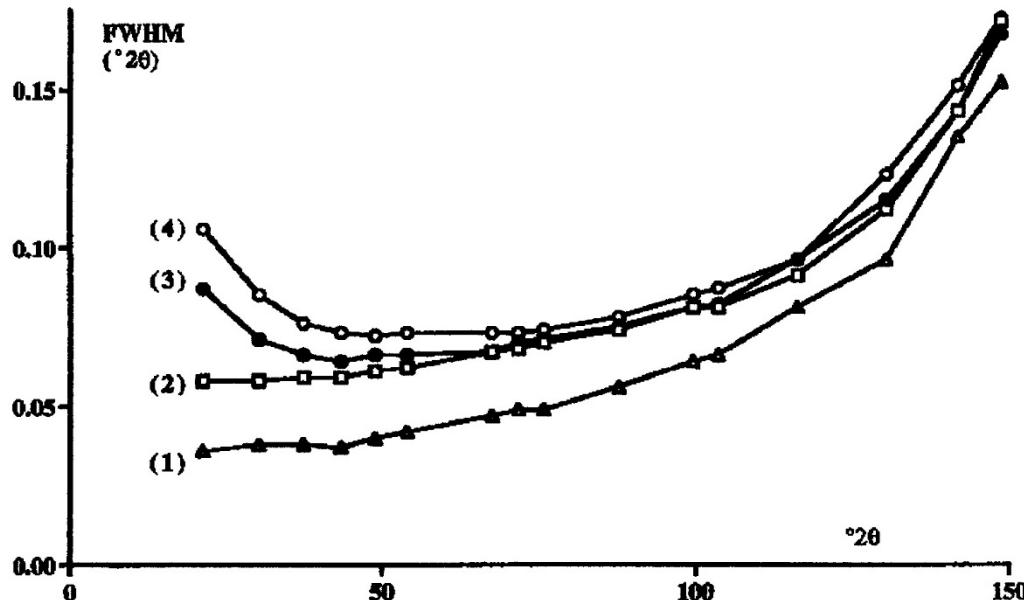


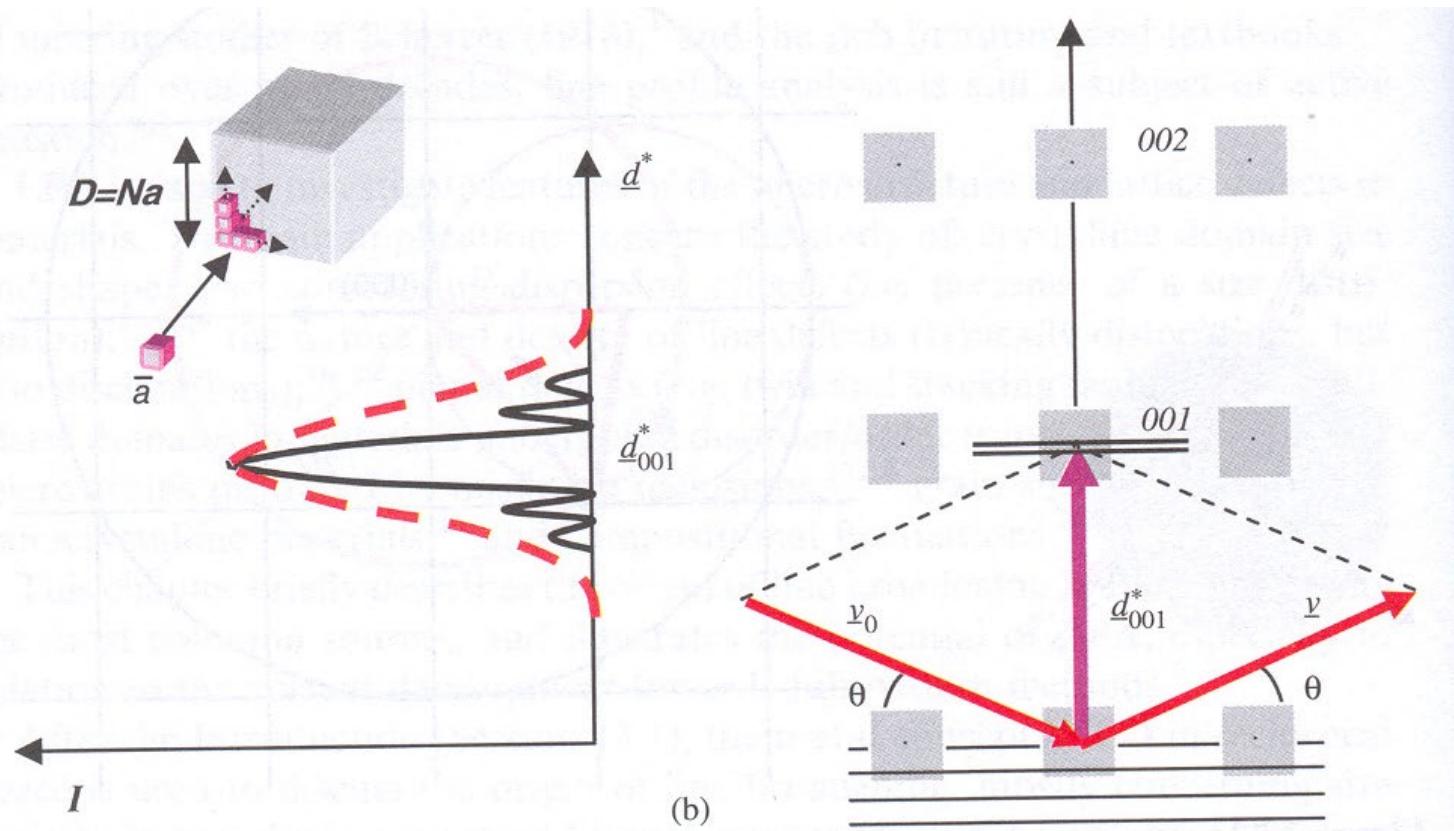
Fig. 3. FWHM curves for SRM 660 LaB6 different slit configurations. Curve (1) corresponds to a receiving slit =  $0.013^\circ$ , a divergence slit of  $0.25^\circ$  and two Soller slits. Each curve corresponds to a change in one of the slits. Curve (1) to (2) is produced by an increase in receiving slit from  $0.013^\circ$  to  $0.053^\circ$ . Curve (2) to (3) arises from the increase in divergence slit from  $0.25^\circ$  to  $1.25^\circ$ . Curve (3) to (4) occurs when the diffracted beam Soller slit is removed. Error bars have been omitted because they are approximately the size of the plotted symbols.

R. W. Cheary and J. P. Cline, "An Analysis of the Effect of Different Instrumental Conditions on the Shapes of X-ray Powder Line Profiles", *Adv. X-ray Anal.*, **38**, 75-82 (1995).

# Specimen Contributions

Size and (micro)Strain

# Size Broadening



**Figure 13.2** Schematic representation of the  $(001)$  diffraction condition (right) and amplitude of the diffracted intensity (left) in reciprocal space for an ideally perfect crystal (a) and for cubic crystalline domains of edge  $D$  [inset of (b)]. The profile for a dispersed system of cubic crystallites (dashed line) is also sketched out in (b).

P. Scardi, “Microstructural Properties: Lattice Defects and Domain Size Effects”, in R. E. Dinnebier and S.J. L. Billinge, *Powder Diffraction: Theory and Practice*, RSC Publishing (2008)

# Integral Breadth

$$\beta(s) = \frac{\int_{-\infty}^{\infty} I(s) ds}{I(0)} = \frac{\int_{-\infty}^{\infty} \frac{\sin^2(\pi Nas)}{(\pi as)^2} ds}{\lim_{s \rightarrow 0} \frac{\sin^2(\pi Nas)}{(\pi as)^2}} = \frac{Na}{(Na)^2} = \frac{1}{D}$$

Convert to  $2\theta$  space from reciprocal space:

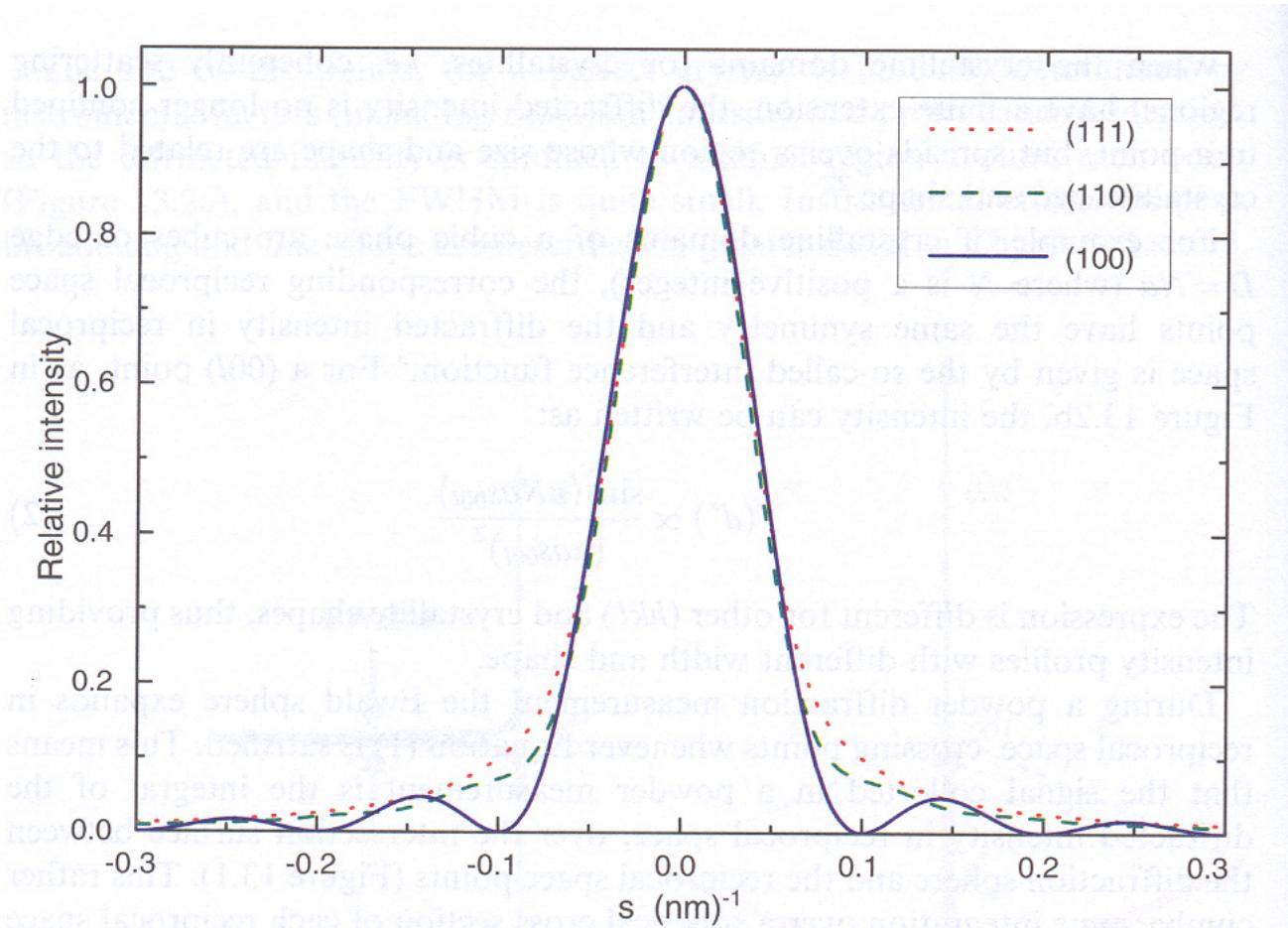
$$\beta(2\theta) = \frac{\lambda K_\beta}{D \cos \theta}$$

# Scherrer Constants for Various Crystallite Shapes

Shape	K (FWHM)	K (integral breadth)
Sphere	0.89	1.07
Cube	0.83-0.91	1.00-1.16
Tetrahedron	0.73-1.03	0.94-1.39
Octahedron	0.82-0.94	1.04-1.14

J. I. Langford and A. J. C. Wilson, “Scherrer after sixty years: A survey and some new results in the determination of crystallite size”, *J. Appl. Cryst.*, **11**, 102-113 (1978)

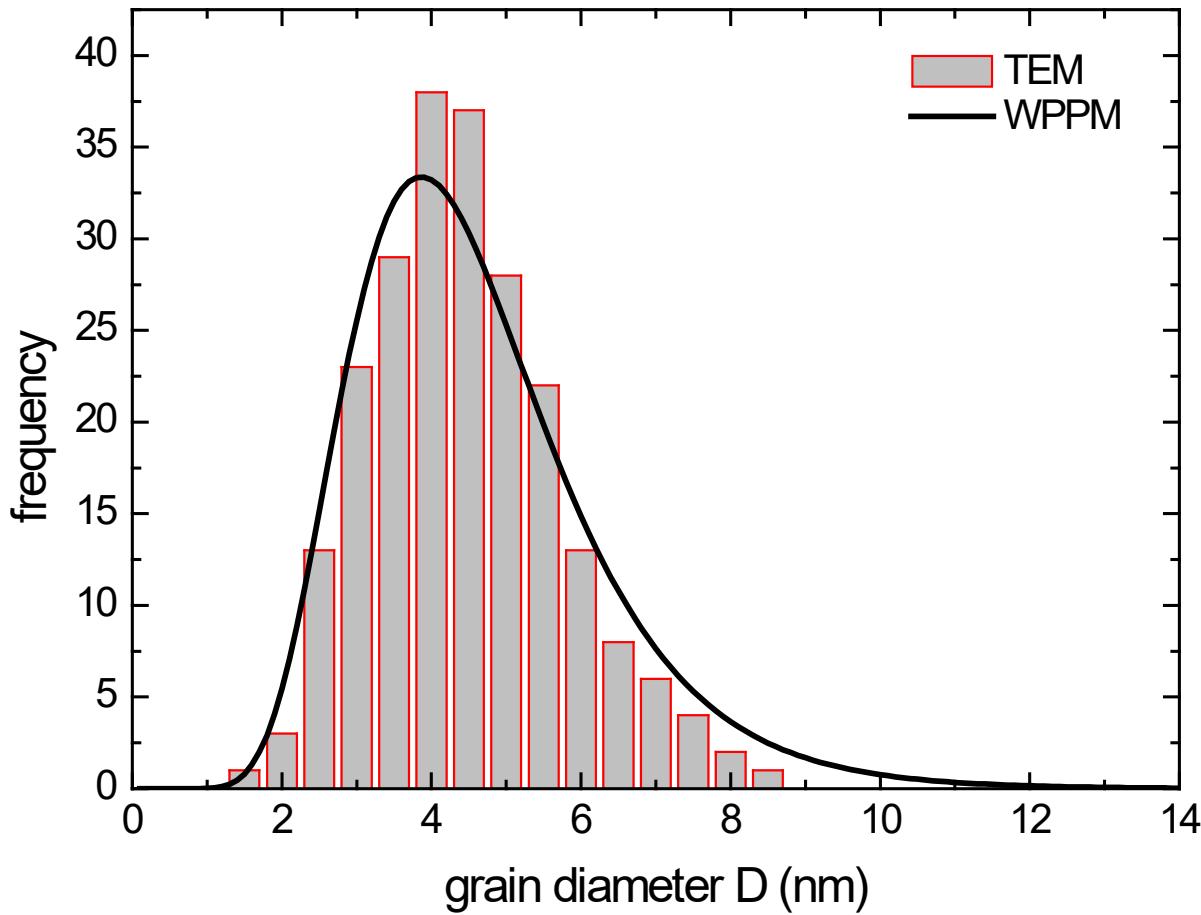
# Shape?



**Figure 13.3** PD (100) (line), (110) (dash) and (111) (dot) peak profiles for a system made of cubic crystallites (edge  $D=10 \text{ nm}$ ). Normalized profiles in reciprocal space.

P. Scardi and M. Leoni,  
“Diffraction line profiles from  
polydisperse crystalline  
systems”, *Acta Cryst. Sect. A*,  
**57**, 604-613 (2001).

# Size Distribution in Ceria Powder



# Strain Broadening

# Macrostrain

$$\lambda = 2d \sin \theta$$

$$0 = 2dd \sin \theta + 2d \cos \theta d\theta$$

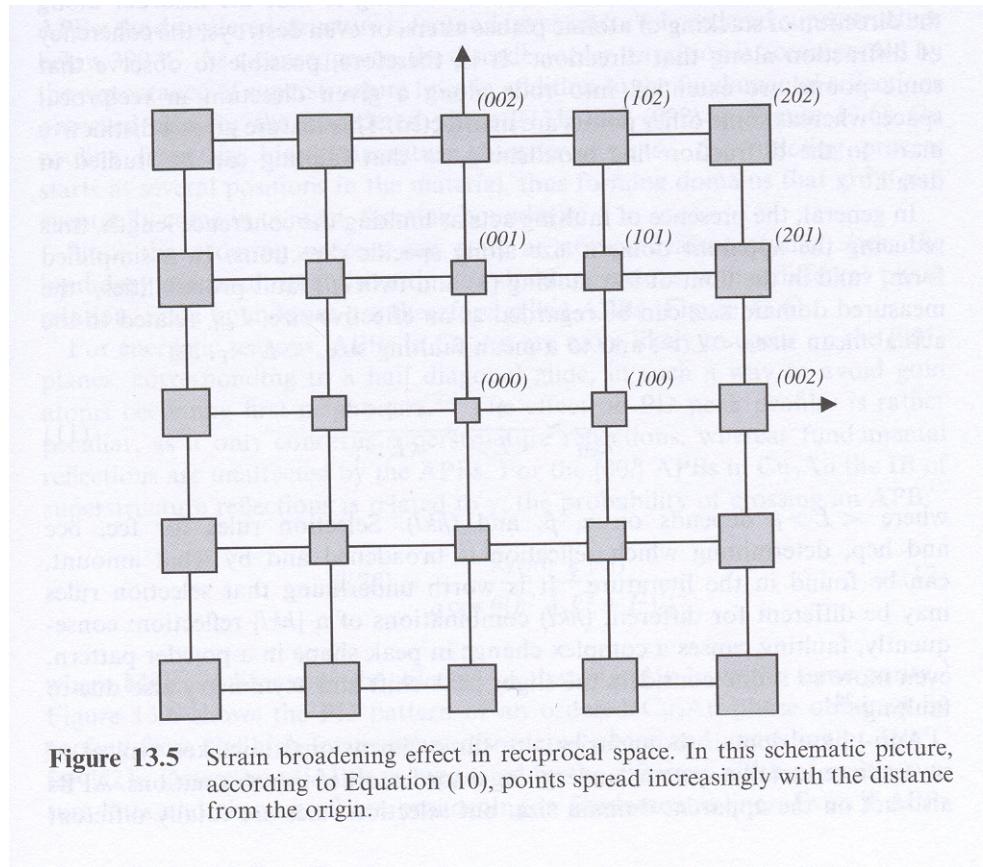
$$0 = 2\Delta d \sin \theta + 2d \cos \theta \Delta \theta$$

$$\Delta 2\theta = -2 \frac{\Delta d}{d} \tan \theta = -2\varepsilon \tan \theta$$

# Microstrain

$$\beta(2\theta) \propto \left\langle \varepsilon^2 \right\rangle^{1/2} \tan \theta$$

# Microstrain



**Figure 13.5** Strain broadening effect in reciprocal space. In this schematic picture, according to Equation (10), points spread increasingly with the distance from the origin.

P. Scardi, “Microstructural Properties: Lattice Defects and Domain Size Effects”, in R. E. Dinnebier and S.J. L. Billinge, *Powder Diffraction: Theory and Practice*, RSC Publishing (2008)

# Anisotropic Strain

P.W. Stephens, “Phenomenological model of anisotropic peak broadening in powder diffraction”, *J. Appl. Cryst.*, **32**, 281-289 (1999)

$$\frac{1}{d^2} = M_{hkl} = Ah^2 + Bk^2 + Cl^2 + Dkl + Ehl + Fhk$$

$$\frac{1}{d^2} = M_{hkl} = \alpha_1 h^2 + \alpha_2 k^2 + \alpha_3 l^2 + \alpha_4 kl + \alpha_5 hl + \alpha_6 hk$$

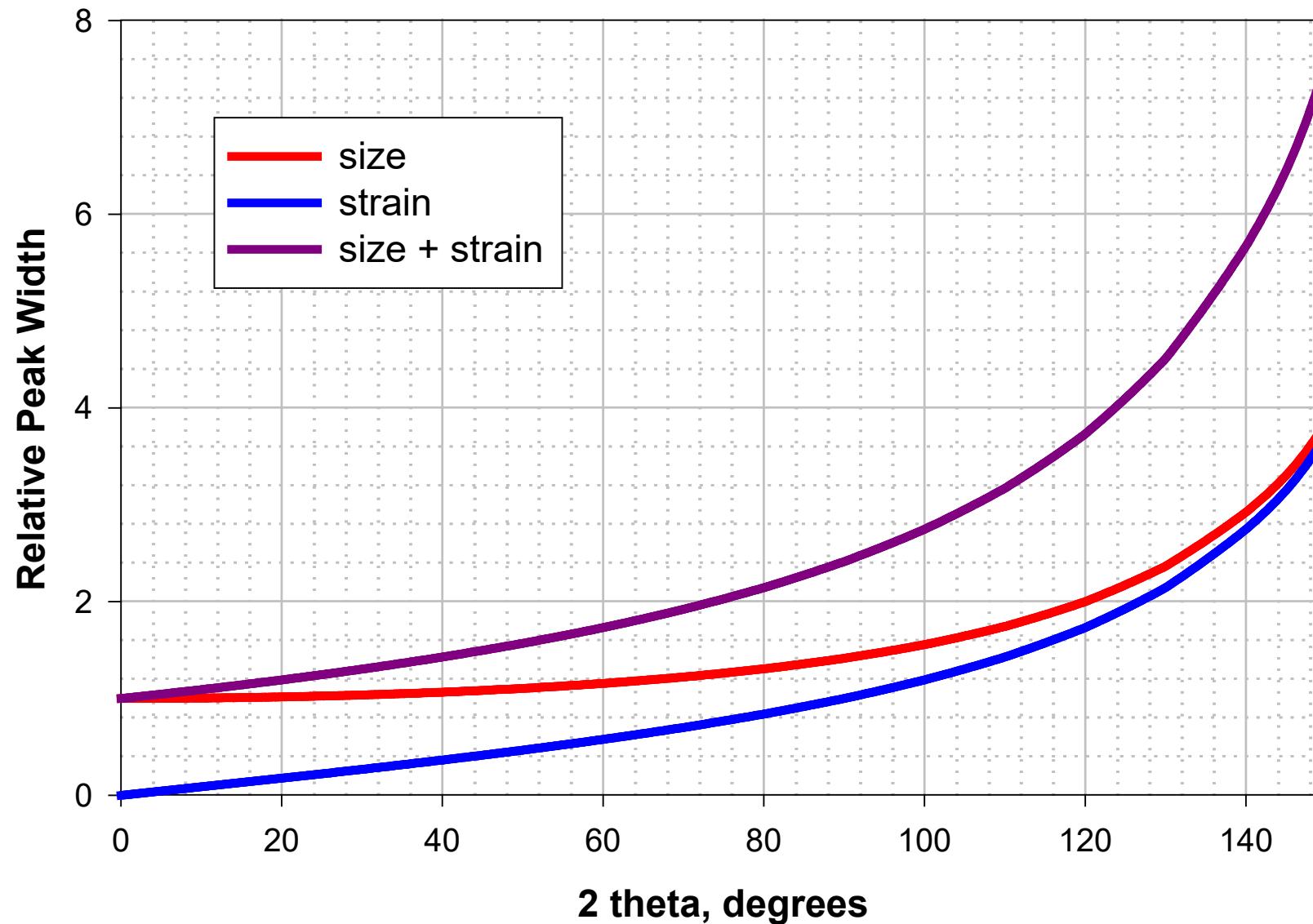
$$\sigma^2(M_{hkl}) = \sum_{i,j} C_{ij} \frac{\partial M}{\partial \alpha_i} \frac{\partial M}{\partial \alpha_j}$$

$$\sigma^2(M_{hkl}) = \sum_{HKL} S_{HKL} h^H k^K l^L$$

$$H + K + L = 4$$

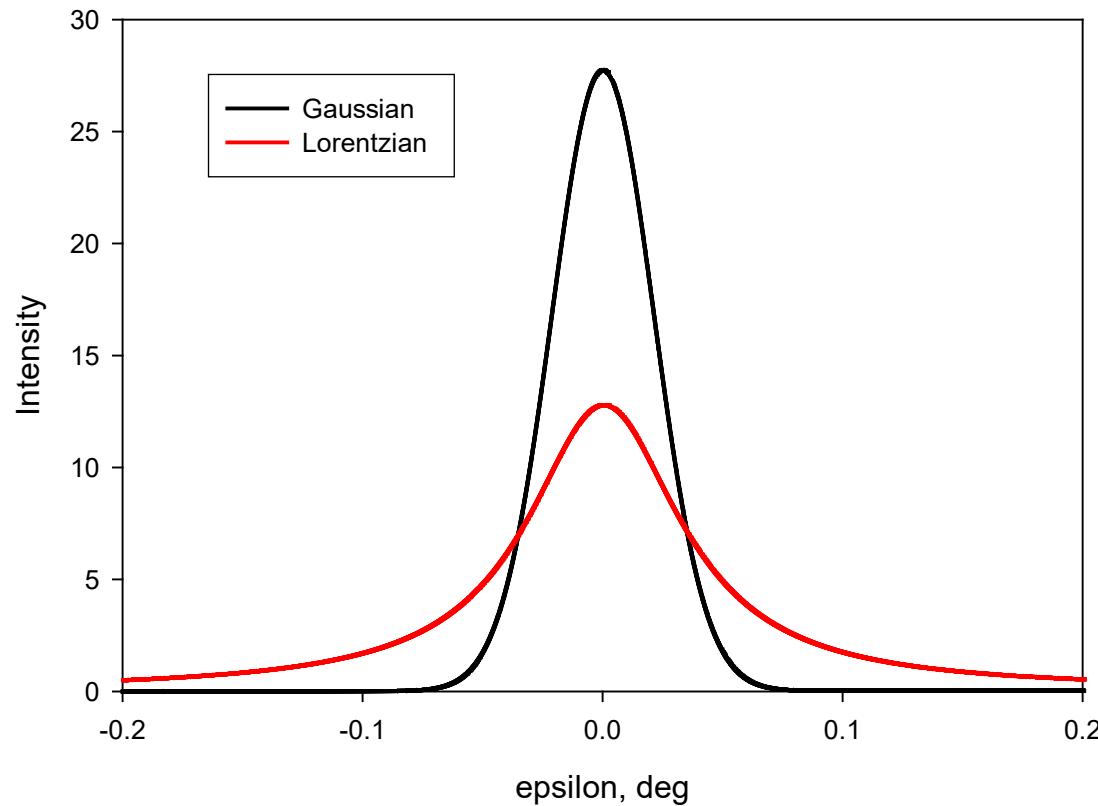
$$\begin{aligned} \sigma^2(M_{hkl}) = & S_{400} h^4 + S_{040} k^4 + S_{004} l^4 + 3(S_{220} h^2 k^2 \\ & + S_{202} h^2 l^2 + S_{022} k^2 l^2) + 2(S_{310} h^3 k + S_{103} h l^3 \\ & + S_{031} k^3 l + S_{130} h k^3 + S_{301} h^3 l + S_{013} k l^3) \\ & + 3(S_{211} h^2 k l + S_{121} h k^2 l + S_{112} h k l^2) \end{aligned}$$

# Functional Forms of Size and Strain Broadening



# Williamson-Hall Analysis

# Real peaks have both Gaussian and Lorentzian (Cauchy) components



Same FWHM and area!

# Profile Equations

Gaussian

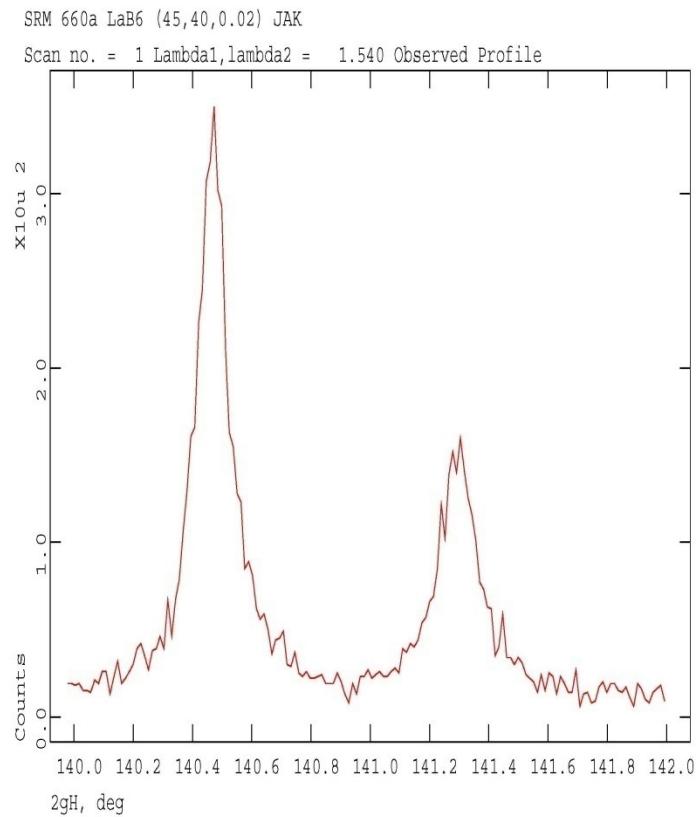
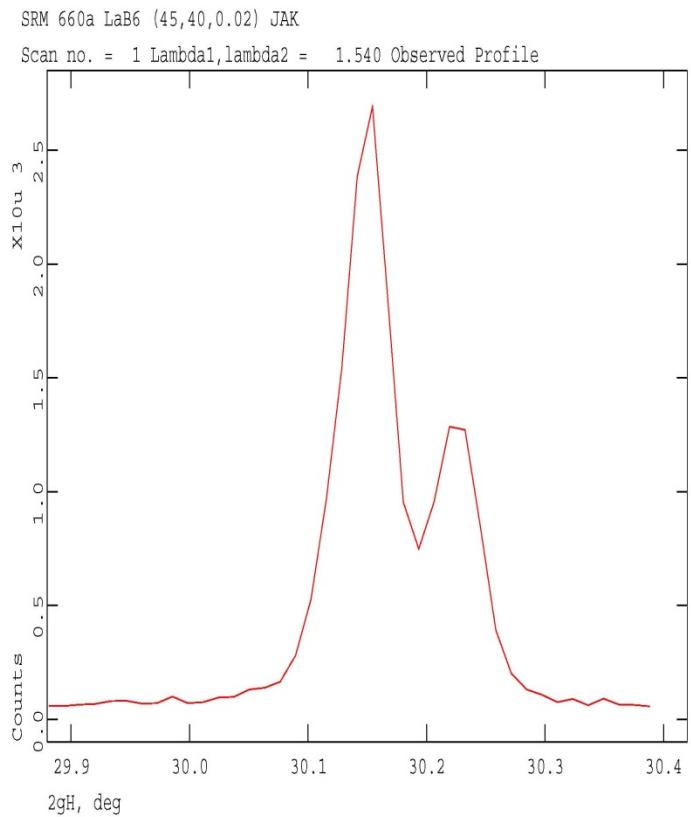
$$I_{i,k} = \frac{2\sqrt{\ln 2}}{H_k} \exp\left[\frac{-4\ln 2}{H_k^2} \varepsilon^2\right]$$

Lorentzian

$$I_{i,k} = \frac{2}{\pi H_k} \left[ 1 + \frac{4(\sqrt{2}-1)}{H_k^2} \varepsilon^2 \right]^{-1}$$

S. A. Howard and K. D. Preston, “Profile Fitting of Powder Diffraction Patterns”,  
in D. L. Bish and J. E. Post, *Modern Powder Diffraction* (1989), p. 217-275.

# SRM 660a LaB<sub>6</sub>



So use combination of  
Gaussian and Lorentzian

Voigt (convolution)  
pseudo-Voigt (sum)

L. W. Finger, D. E. Cox, and A. P. Jephcoat, "A Correction for Powder Diffraction Peak Asymmetry due to Axial Divergence",  
*J. Appl. Cryst.*, **27**, 892-900 (1994).

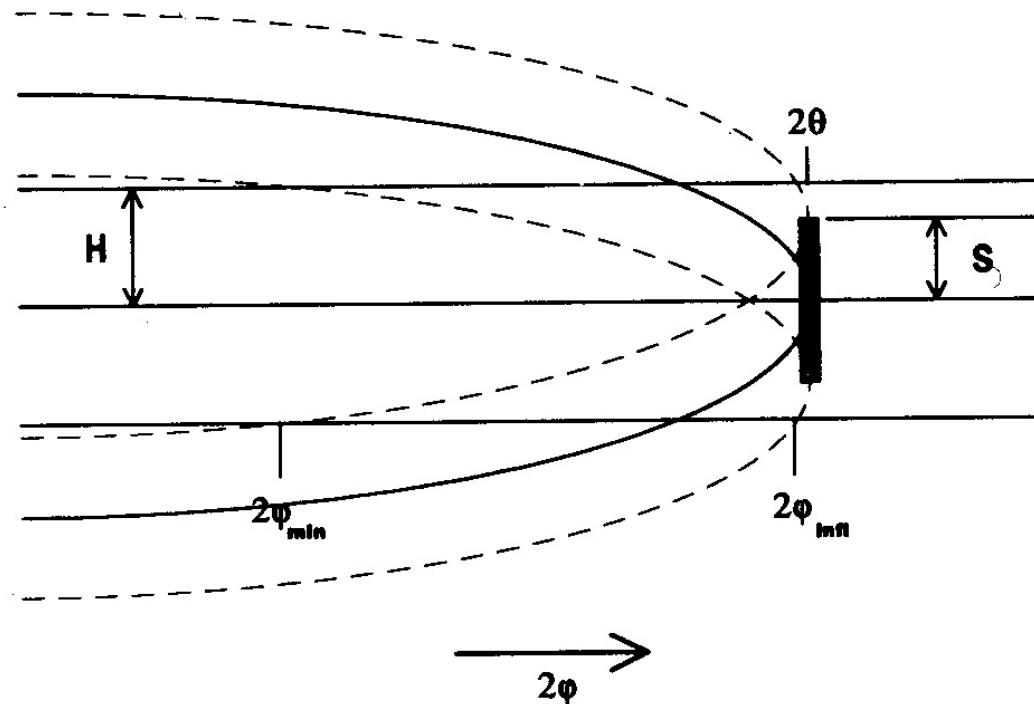


Fig. 1. The band of intensity, diffracted by a sample with height  $2S$ , as seen by a detector with opening  $2H$  and a detector angle  $2\varphi$  moving in the detector cylinder. The figure is adapted from that of van Laar & Yelon (1984). For angles below  $2\varphi_{\min}$ , no intensity is seen. For angles between  $2\varphi_{\min}$  and  $2\theta$ , the entire sample can be seen by the detector.

# GSAS-I Profile Functions #3-5

S/L = sample “half height”/diffractometer radius

H/L = slit “half height”/diffractometer radius

$$6/240 = 0.025$$

# GSAS-I Profile Function #2 (3-5)

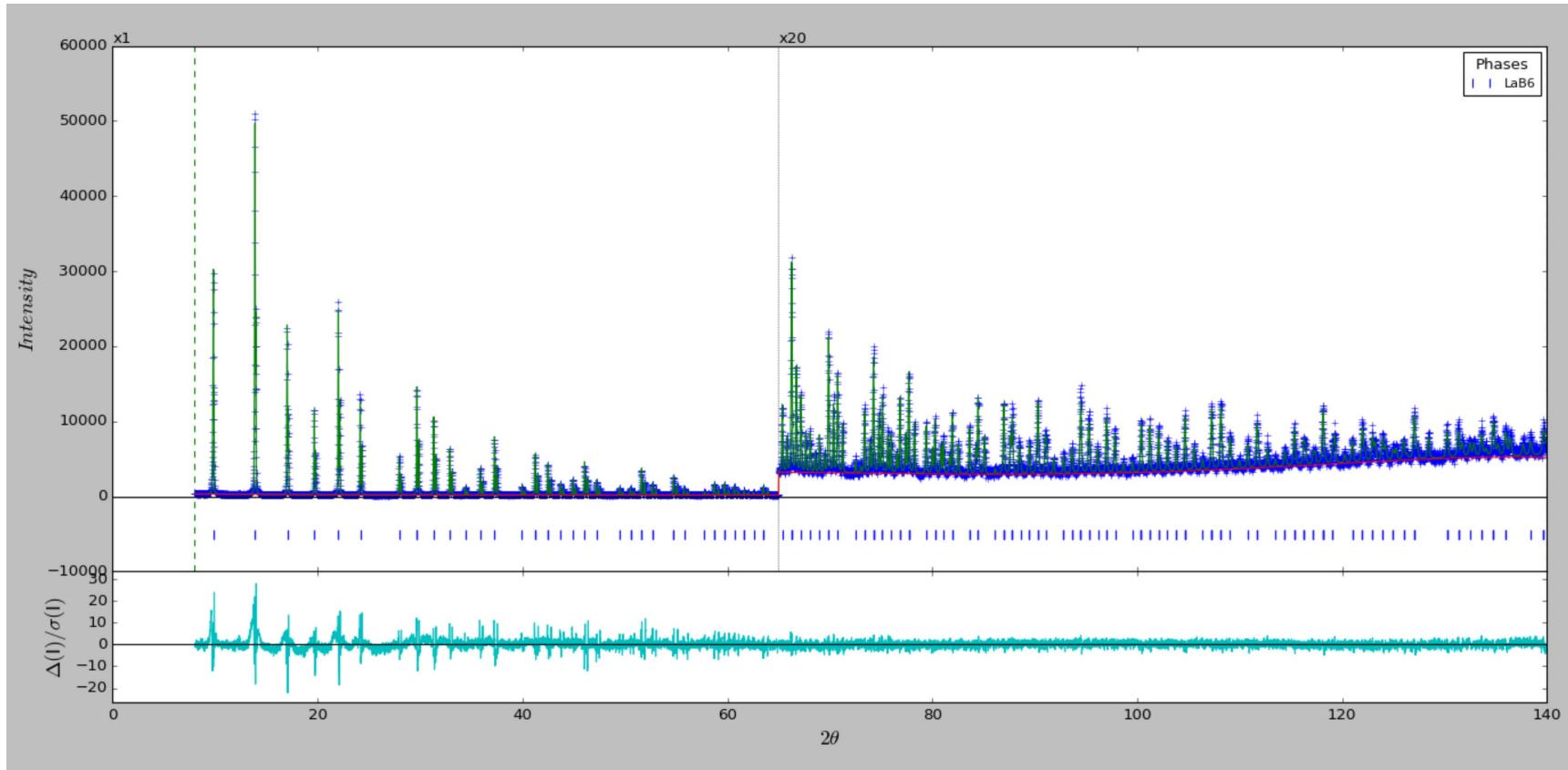
## Thompson-Cox-Hastings pseudo-Voigt

$$\sigma^2 = GU \tan^2 \theta + GV \tan \theta + GW + \frac{GP}{\cos^2 \theta}$$

$$\gamma = \frac{LX + ptec \cos \varphi}{\cos \theta} + (LY + stec \cos \varphi) \tan \theta$$

$$\Delta 2\theta = zero + \left( \frac{f_i asym}{\tan 2\theta} \right) + shft \cos \theta + trns \sin 2\theta$$

# NIST SRM 660c LaB<sub>6</sub>



# GSAS-II

GSAS-II project: kadu1814.gpx

File Data Calculate Import Export | Operations | Help

Project: C:\zjak01\NCC\kadu1814

- Notebook
- Controls
- Covariance
- Constraints
- Restraints
- Rigid bodies
- PWDR kadu1814.gdas Bank 1
  - Comments
  - Limits
  - Background
  - Instrument Parameters**
  - Sample Parameters
  - Peak List
  - Index Peak List
  - Unit Cells List
  - Reflection Lists
- Phases

Histogram Type: PXC Bank: 1  
Azimuth: 0.00 Ka1/Ka2: 0.709260/0.713543Å Source type: MoKa

Name (default)	Value	Refine?
I(L2)/I(L1) (0.5800):	0.5	<input type="checkbox"/>
Zero (0.0000):	0.0653	<input checked="" type="checkbox"/>
Polariz. (0.5000):	0.5	<input type="checkbox"/>
U (33.990):	28.116	<input checked="" type="checkbox"/>
V (0.000):	0.0	<input type="checkbox"/>
W (1.704):	1.623	<input checked="" type="checkbox"/>
X (0.000):	0.487	<input checked="" type="checkbox"/>
Y (0.000):	0.0	<input type="checkbox"/>
Z (0.000):	0.0	<input type="checkbox"/>
SH/L (0.00200):	0.03481	<input checked="" type="checkbox"/>

< >

Mouse RB drag/drop to reorder NB: Azimuth is used for polarization only

# GSAS-II Instrument Parameter File

## PANalytical Empyrean/Mo/capillary

```
#GSAS-II instrument parameter file; do not add/delete items!
```

```
I(L2)/I(L1):0.5
```

```
SH/L:0.0348141083102
```

```
Azimuth:0.0
```

```
Lam2:0.713543
```

```
Source:MoKa
```

```
Zero:0.0652623352976
```

```
Lam1:0.70926
```

```
U:28.1161839663
```

```
W:1.62280279963
```

```
V:0.0
```

```
Y:0.0
```

```
X:0.486972245354
```

```
Z:0.0
```

```
Type:PXC
```

```
Bank:1
```

```
Polariz.:0.5
```

# GSAS-II Phase Data

The image displays two windows of the GSAS-II software:

**Instrument Parameters Window:**

- File Data Calculate Import Export | Operations | Help
- Project: C:\MyFiles\DXC\2017\
- Histogram Type: PXC Bank: 1
- Name (default) Value Refine?
- Azimuth: 45.81
- Lam (Å): 0.452410
- Zero (0.000): 0.0
- Polariz. (0.990): 0.99
- U (21.773): 107.52
- V (27.634): 0.0
- W (12.953): 15.35
- X (0.00): 0.102
- Y (0.00): 0.0
- SH/L (0.00200): 0.0005

**Phase Data for Cu O Window:**

- File Data Calculate Import Export | Select tab Edit Phase | Help
- Project: C:\MyFiles\DXC\2017\
- General Data Atoms Draw Options Draw Atoms RB Models Map peaks M...
- Histogram data for Cu O:
- Select plot type:
  - None (radio button selected)
  - Mstrain
  - Size
  - Preferred orientation
  - Inv. pole figure
- Use Histogram: PWDR CuO\_red\_CO\_300C-00000.tif\_A180.fxye Bank 1?  Do LeBail extraction?
- Phase fraction: 1.0 Wt. fraction: 1.000
- Domain size model: isotropic  LGmix 1.0000  Reset?
- size(um): 1.000
- Mstrain model: generalized  LGmix 1.0000  Reset?
  - S400 500674.9  S040 4453211.1  S004 165105.6
  - S220 -202901.7  S202 672136.7  S022 1158429.4
  - S301 55943.0  S103 423215.2  S121 736375.8
- Hydrostatic/elastic strain:
  - D11 0  D22 0  D33 0
  - D13 0
- Preferred orientation model: March-Dollase 
  - March-Dollase ratio: 1.0 Unique axis, H K L: 0 0 1
  - Extinction: 0.0
  - Babinet A: 0.0  Babinet U: 0.0

# Control of Peak Positions

- Lattice parameters
- Specimen displacement
- Specimen transparency
- (Zero)

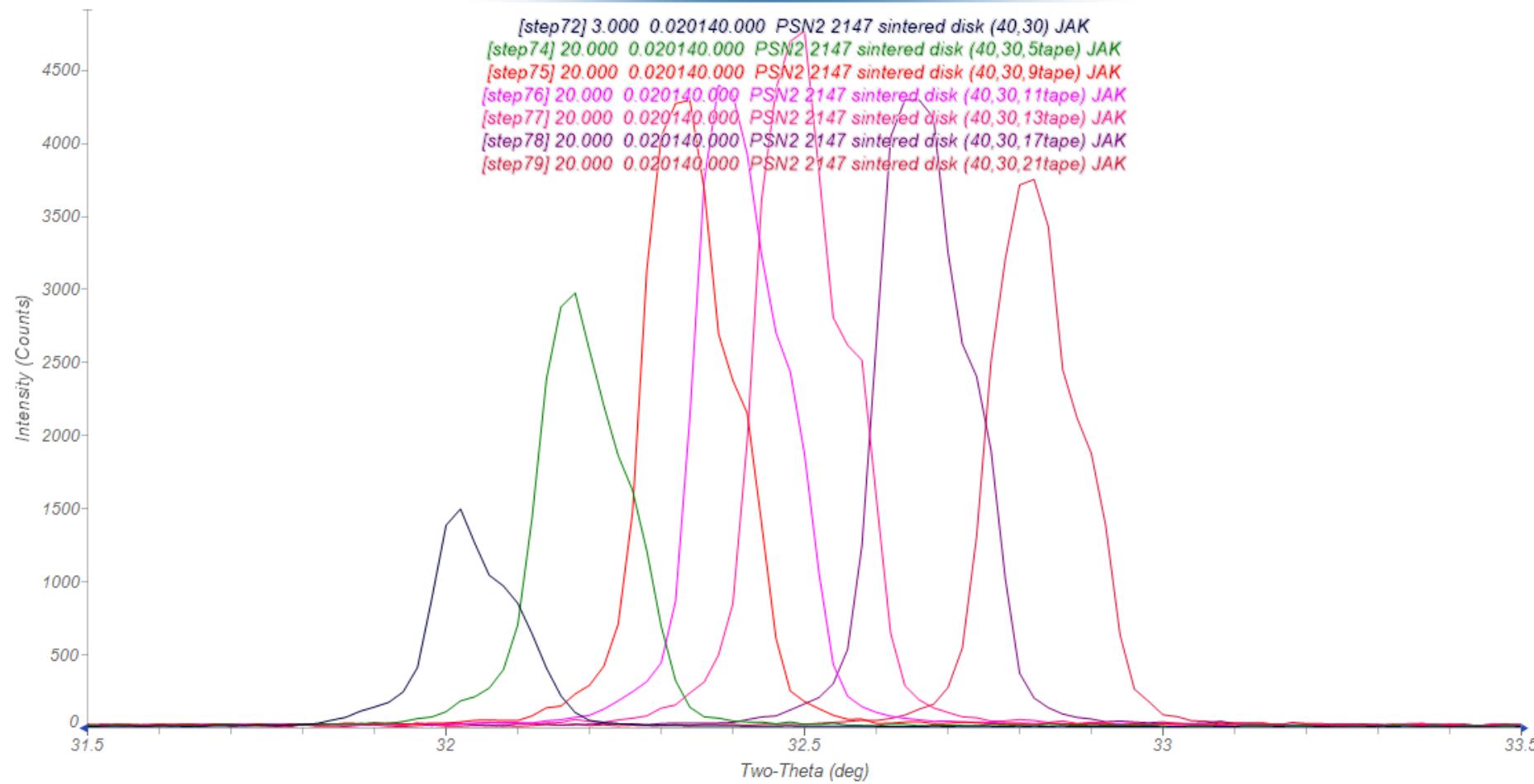
# Specimen Displacement (GSAS-I)

$$\Delta 2\theta = shft \cos \theta$$

$$s = \frac{-\pi R shft}{36000}$$

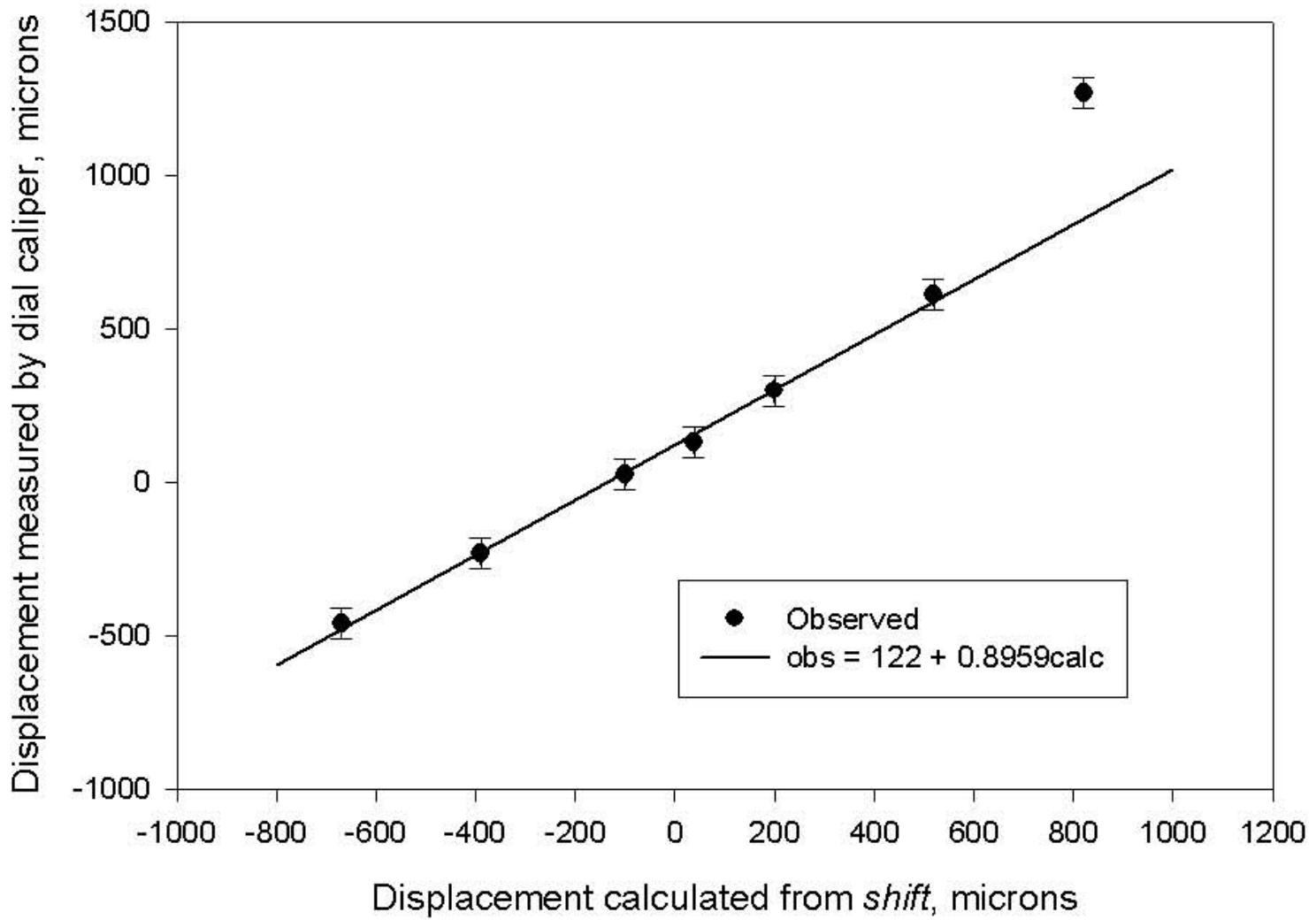
[step72] 3.000 0.020140.000 PSN2 2147 sintered disk (40,30) JAK

[step72] 3.000 0.020140.000 PSN2 2147 sintered disk (40,30) JAK  
[step74] 20.000 0.020140.000 PSN2 2147 sintered disk (40,30,5tape) JAK  
[step75] 20.000 0.020140.000 PSN2 2147 sintered disk (40,30,9tape) JAK  
[step76] 20.000 0.020140.000 PSN2 2147 sintered disk (40,30,11tape) JAK  
[step77] 20.000 0.020140.000 PSN2 2147 sintered disk (40,30,13tape) JAK  
[step78] 20.000 0.020140.000 PSN2 2147 sintered disk (40,30,17tape) JAK  
[step79] 20.000 0.020140.000 PSN2 2147 sintered disk (40,30,21tape) JAK



# Measured and Calculated Specimen Displacements

## PSN2 2147



# Specimen Displacement

Displacement, $\mu\text{m}$	<i>shift</i>	$a$ , Å
-670	36.79(3)	3.90997(2)
-390	21.25(5)	3.91010(2)
-100	5.56(4)	3.91016(2)
38	-2.06(3)	3.91025(1)
200	-10.78(4)	3.91007(2)
520	-28.57(4)	3.91025(2)
820	-44.75(5)	3.91025(2)
Average		3.9102(1)

# GSAS-II project: kadu937.gpx

File Data Calculate Import Export | Command | Help

- Project: C:\MyFiles\ICDD\_Rietveld
- Notebook
- Controls
- Covariance
- Constraints
- + Restraints
- Rigid bodies
- PWDR kadu937.gsas Bank 1
- Comments
- Limits
- Background
- Instrument Parameters
- Sample Parameters
- Peak List
- Index Peak List
- Unit Cells List
- Reflection Lists
- Phases
- Calcium Tartrate Tetrahy

Sample and Experimental Parameters

Instrument Name	
Diffractometer type:	Bragg-Brentano
<input checked="" type="checkbox"/> Histogram scale factor:	806.94
Goniometer radius (mm):	240.
<input checked="" type="checkbox"/> Sample displacement(μm):	62.9518
<input type="checkbox"/> Sample transparency(1/μeff, cm):	0.0
<input type="checkbox"/> Surface roughness A:	0.0
<input type="checkbox"/> Surface roughness B:	0.0
Goniometer omega:	0.
Goniometer chi:	0.
Goniometer phi:	0.
Detector azimuth:	0.
Clock time (s):	0.
Sample temperature (K):	300.
Sample pressure (MPa):	0.1
Sample humidity (%)	0.
Sample voltage (V)	0.
Applied load (MN)	0.

# GSAS-II project: kadu1814.gpx

File Data Calculate Import Export | Command | Help

Project: C:\zjak01\NCC\kadu181

Notebook

Controls

Covariance

Constraints

Restraints

Rigid bodies

PWDR kadu1814.gsas Bank 1

Comments

Limits

Background

Instrument Parameters

Sample Parameters

Peak List

Index Peak List

Unit Cells List

Reflection Lists

Phases

## Sample and Experimental Parameters

Instrument Name NCC Empyrean

Diffractometer type: Debye-Scherrer

<input checked="" type="checkbox"/> Histogram scale factor:	480.98
Goniometer radius (mm):	240.
<input checked="" type="checkbox"/> Sample X displ. perp. to beam ( $\mu\text{m}$ ):	176.121
<input checked="" type="checkbox"/> Sample Y displ.    to beam ( $\mu\text{m}$ ):	-6.558
<input type="checkbox"/> Sample absorption ( $\mu\text{-r}$ ):	1.75
Goniometer omega:	0.
Goniometer chi:	0.
Goniometer phi:	0.
Detector azimuth:	0.
Clock time (s):	0.
Sample temperature (K):	300.
Sample pressure (MPa):	0.1
Sample humidity (%)	0.
Sample voltage (V)	0.
Applied load (MN)	0.

< >

Mouse RB drag/drop to reorder

# Specimen Transparency

When the specimen is long enough to intercept the whole beam, and

$$t \geq \frac{3.2}{\mu} \frac{\rho}{\rho'} \sin \theta$$

an additional component of the profile  $g$  is generated:

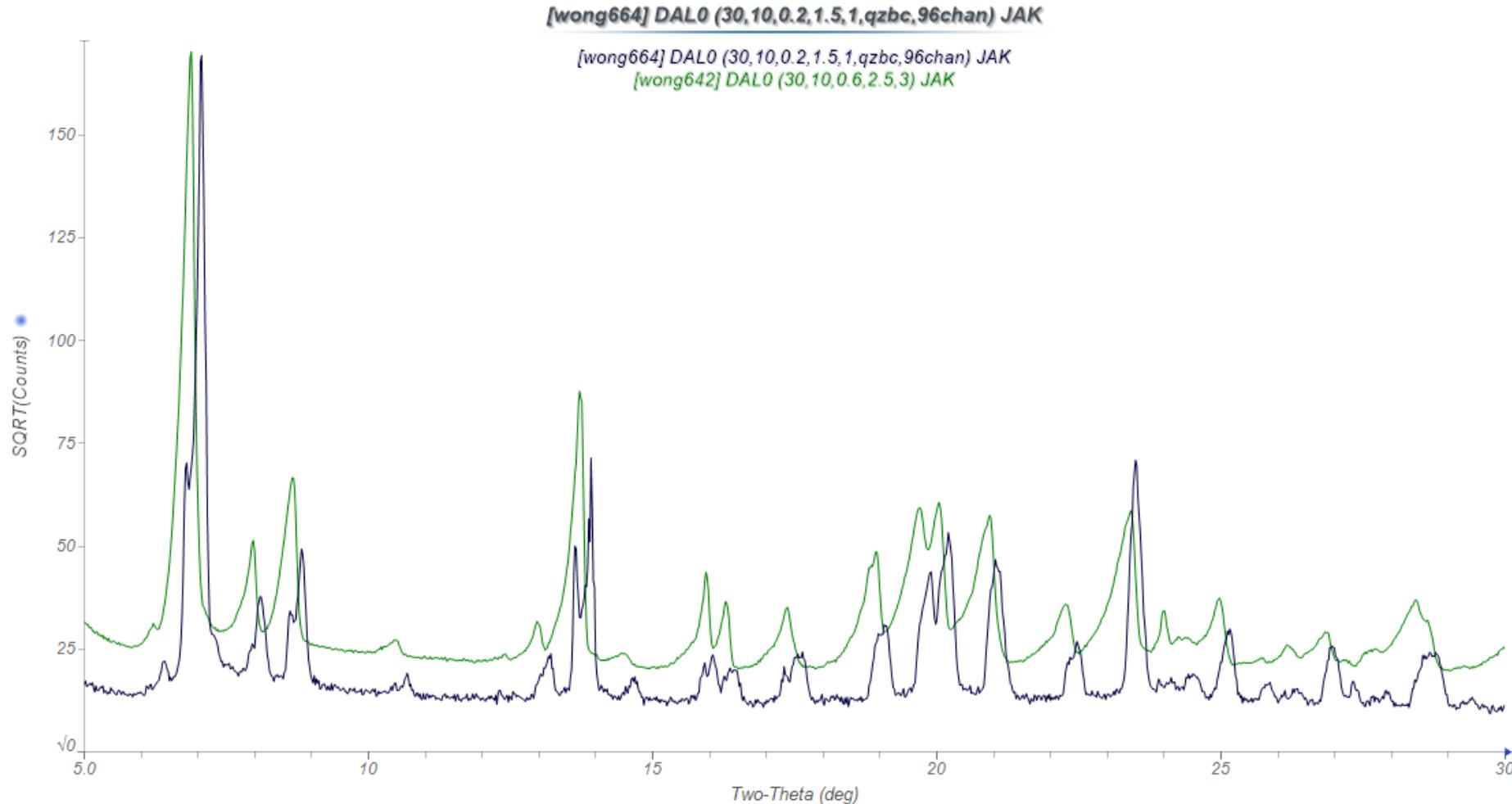
$$g = \exp\left(\frac{4\pi R\varepsilon}{114.6} \sin 2\theta\right)$$

$$-\infty < \varepsilon \leq 0 \text{ } (^{\circ})$$

# Transparency

- Significant for thick organic specimens
- Additional low-angle asymmetry
- Peak shift to low angles

# Extra Low-Angle Asymmetry Peak Shift to Low-Angles



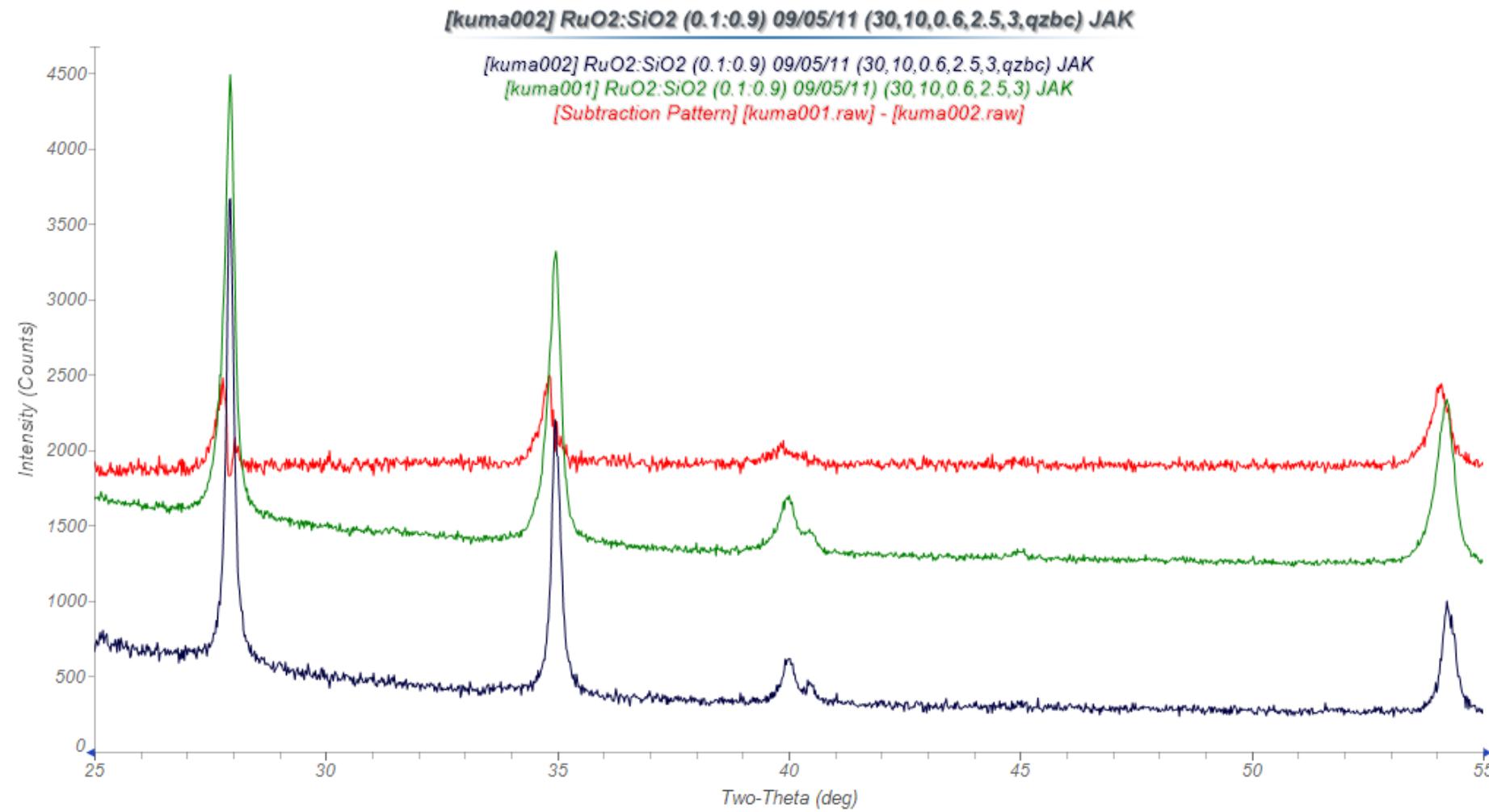
# Specimen Transparency

## GSAS-I

$$\Delta 2\theta = trns \sin 2\theta$$

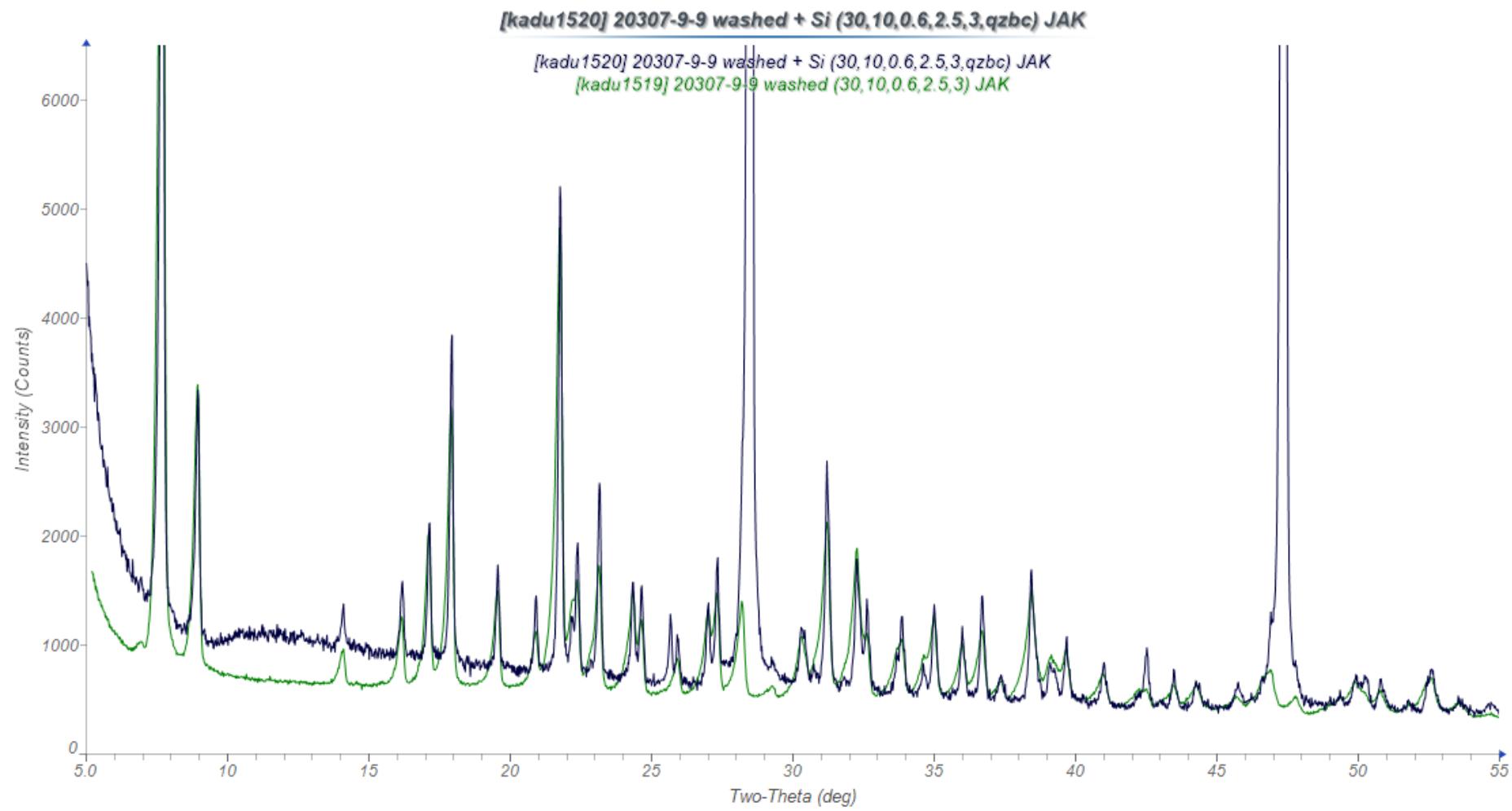
$$\mu_{eff} = \frac{-9000}{\pi R trns}$$

# 10% RuO<sub>2</sub>/SiO<sub>2</sub> Catalyst



# Penetration Depth, $\mu\text{m}$

$2\theta, {}^\circ$	28	130
Pure $\text{RuO}_2$	22	70
10% $\text{RuO}_2$ /90% $\text{SiO}_2$	100	340



# Instrument Profiles

“Typical values of Rietveld instrument profile coefficients”,  
J. A. Kaduk and J. Reid, *Powder Diffraction*, **26**(1), 88-93 (2011).

Table II. GSAS Function #2 Instrument Profile Parameters for a Variety of Laboratory Diffractometers.

Diffractometer	Date	<i>U</i>	<i>V</i>	<i>W</i>	<i>X</i>	<i>Y</i>	<i>asym</i>
X'Pert Pro PIXcel/0.04 rad Soller	08/2010	0.8048	0	0.5103	2.537	1.946	4.343
D2/Lynxeye	05/2010	1.371	0	2.393	2.183	1.199	2.774
D2/Lynxeye	10/2009	2.8329	0	2.695	1.853	2.488	2.194
X'Pert Pro PIXcel/mono	01/2008	0.7565	0	3.646	2.428	1.902	1.063
X'Pert Pro PIXcel/no mon	01/2008	2.6369	0	0	2.778	0	2.486
D8/VANTEC	04/2004	0.2879	0	1.124	2.477	2.103	2.052
PAD V	06/2007	1.0270	0	6.640	1.237	2.693	2.109
D/MAX-B	06/2002	0.567	0	18.680	2.301	1.960	6.048
Miniflex	09/2001	5.568	0	20.47	3.614	0	5.487
PWI7xx	08/1998	0	0	5.217	0	9.77	7.603

Table III. GSAS Function #3 Instrument Profile Parameters for a Variety of Laboratory Diffractometers.

Diffractometer	<i>U</i>	<i>V</i>	<i>W</i>	<i>X</i>	<i>Y</i>	<i>S/L</i>	<i>H/L</i>
X'Pert Pro PIXcel/0.04 rad Soller	1.423	0	0.5061	2.842	1.509	0.03547	0.00522
D2/Lynxeye	1.376	0	2.640	2.410	0.850	0.02951	0.0005
X'Pert Pro PIXcel/mono	1.153	-0.928	4.161	2.472	1.814	0.01577	0.0005
X'Pert Pro PIXcel/no mon	2.314	0	0	3.040	0	0.02788	0.0005
D8/VÅNTEC	0.3365	0	1.032	2.526	2.051	0.02695	0.0005
PAD V	1.103	0	6.412	1.173	2.842	0.03018	0.0005
D/MAX-B	3.219	-7.822	24.370	2.460	1.609	0.03858	0.0005

In all of these profile functions,  $P=0$ .

Table IV. GSAS Profile #2 Functions  
for Several Synchrotron Diffractometers

Instr.	Date	<i>U</i>	<i>V</i>	<i>W</i>	<i>X</i>	<i>Y</i>	<i>asym</i>
APS 5-BM-C	10/2002	0.1	0	0	0.2505	0.9462	0
APS 5-BM-C	08/2006	17.1	-8.8	1.3	0	0	0
APS 1-ID	02/2002	0.1	0	0	0.2505	0.9462	0.0646
APS 10-ID-B	01/2000	0.3540	0	0.2908	0.3565	0.5177	0.4744
APS 32-ID	12/2004	0.3120	0	0.0104	0.1186	0.4062	0.0419
LNLS D10B		0.8777	-0.1600	0.1063	0.7604	1.1904	0.5157
NSLS X3B1	03/2004	6.427	-1.067	0	0.6102	0.6796	0.6733

Table V. GSAS Profile #3 Instrument Parameters  
for Several Synchrotron Diffractometers

<b>Inst.</b>	<b>Date</b>	<b><i>U</i></b>	<b><i>V</i></b>	<b><i>W</i></b>	<b><i>P</i></b>	<b><i>X</i></b>	<b><i>Y</i></b>	<b><i>S/L</i></b>	<b><i>H/L</i></b>
APS 5-BM-C	10/2002	1.212	0	0	0	1.980	0	0.00135	0.00718
APS 1-ID	02/2002	0.1	0	0	0	0.1845	11.190	0.0005	0.00458
APS 10-ID	10/2003	1.212	0	0	0	0.198	0	0.00135	0.00718
APS 11-BMB	02/2009	1.163	-0.126	0.063	0	0.173	0	0.00110	0.00110
APS 32-ID	12/2004	1.212	0	0	0	0.198	0	0.00135	0.00718
AS PD		0.0522	0.5640	0.0621	0	0.293	0.171	0.0000	0.0000
NSLS X7B		0	-125.9	73.3	0	2.03	0	0.0001	0.1000
NSLS X16C		0	0	0	1	3	30	0.014	0.014

# GSAS/FullProf Conversions

$$GU(\text{GSAS}) = 1803.4U(\text{FullProf}) \quad (9)$$

$$GV(\text{GSAS}) = 1803.4V(\text{FullProf}) \quad (10)$$

$$GW(\text{GSAS}) = 1803.4W(\text{Fullprof}) \quad (11)$$

$$GP(\text{GSAS}) = 1803.4IG(\text{FullProf}) \quad (12)$$

$$LX(\text{GSAS}) = 100Y(\text{FullProf}) \quad (13)$$

$$LY(\text{GSAS}) = 100X(\text{Fullprof}) \quad (14)$$

$$S/L(\text{GSAS}) = S\ L(\text{Fullprof}) \quad (15)$$

$$H/L(\text{GSAS}) = \overline{D}\ L(\text{Fullprof}) \quad (16)$$

# GSAS-II

## Instrument Parameters/Operations/Save Profile

# More on Size and Strain

Whole powder pattern modelling: microstructure determination from powder data  
Domain size and domain-size distributions  
Stress and Strain



Chapters 3.6 (Leoni), 5.1 (Leoni),  
and 5.2 (Popa) in *International  
Tables for Crystallography Volume  
H: Powder Diffraction* (2019).

# Williamson-Hall Analysis

$$\beta(d^*) = \frac{K_\beta}{\langle D \rangle} + 2ed^*$$

Integral breadth  
Volume-weighted average  
Lorentzian profiles

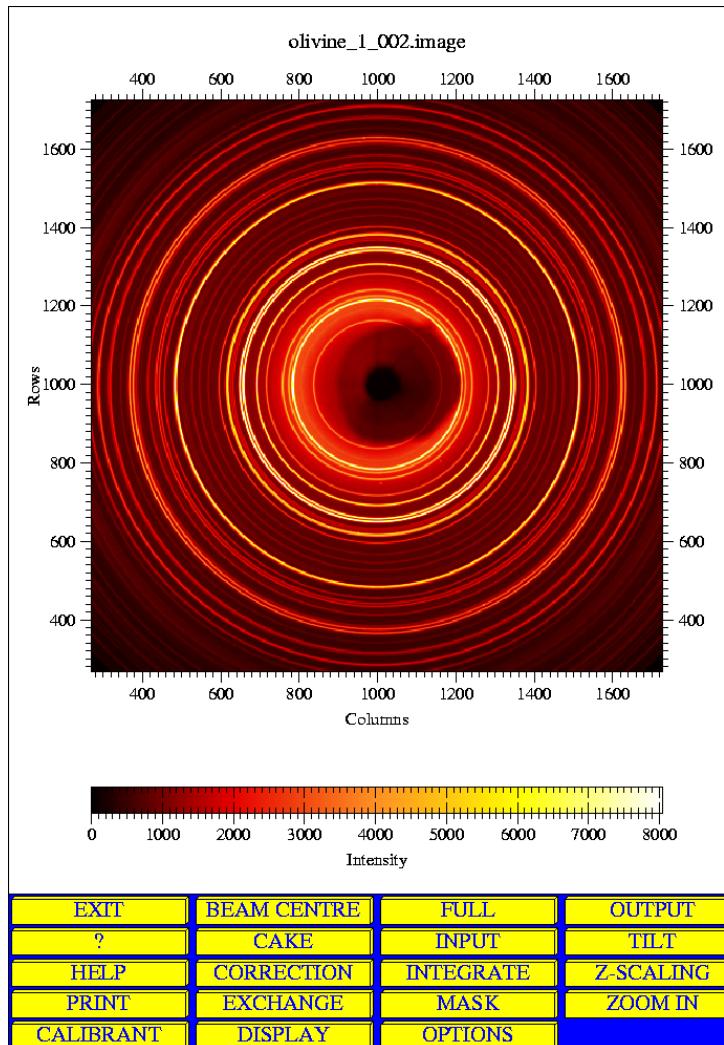
# Fourier Methods (Warren-Averbach, WPPM)

$$h(s) = \int_{-\infty}^{\infty} f(y)g(s - y) dy$$

$$s = d^* - d_{hkl}^* = \frac{2}{\lambda}(\sin\theta - \sin\theta_{hkl})$$

$$FT[h(s)] = FT[f(s)] \times FT[g(s)]$$

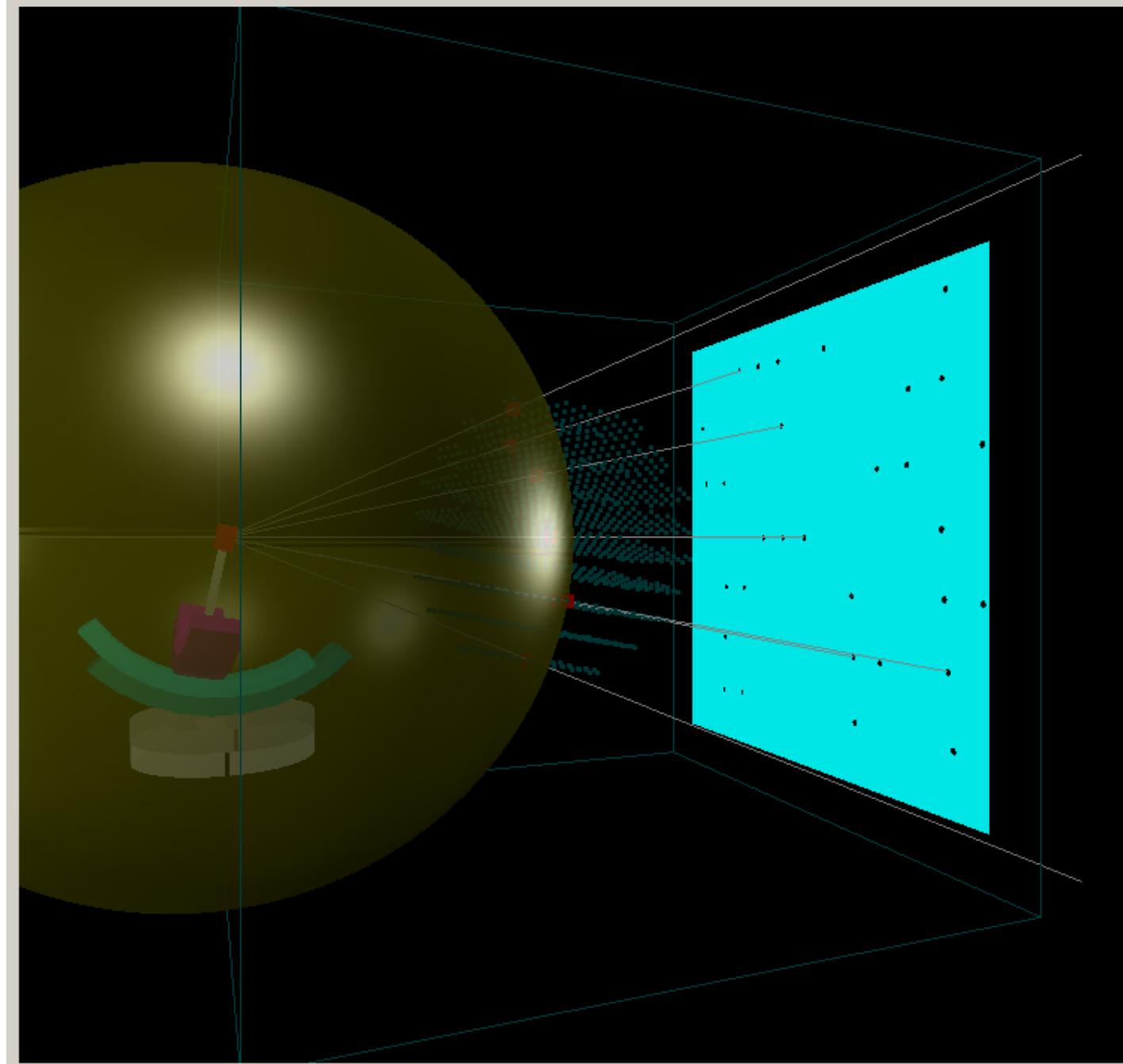
# Texture (Preferred Orientation)



# Quantitative texture analysis and combined analysis



D. Chateigner, L. Lutterotti, and M. Morales, Chapter 5.3 in *International Tables for Crystallography Volume H: Powder Diffraction* (2019).



## GENERAL CONTROLS

## Eye Point Control

Y : -21  
[◀] [■] [▶] Center

X : 0  
[◀] [■] [▶] Center

out... Zoom : 0 % ...in  
[◀] [■] [▶] Center

## Goniometer Controls

Small : 0.00  
[◀] [■] [▶] Center

Large : -11.50  
[◀] [■] [▶] Center

Phi : 4.90  
[◀] [■] [▶] Center

## Other Controls

Auto Rotation

Laue Photography

Lattice Type : P

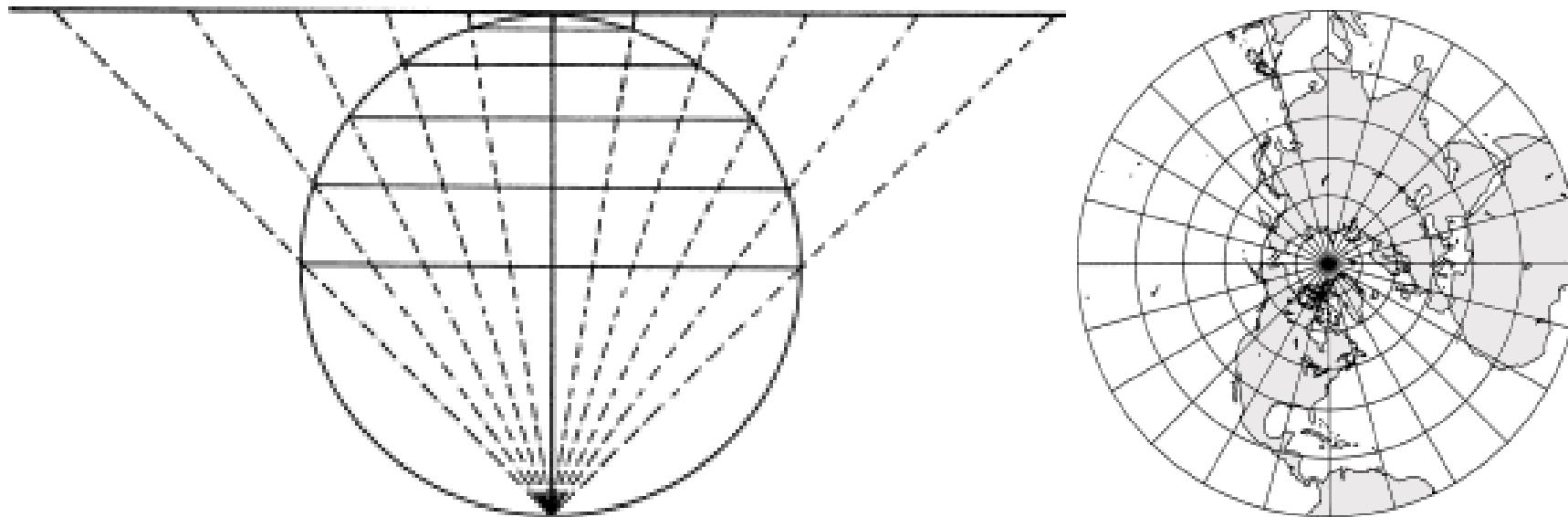
Crystal to Detector Distance : 1.00  
[◀] [■] [▶] Center

Toggle View

Toggle Integration

## OPTIONS CONTROLS

# Stereographic Projection



<http://www.3dsoftware.com/Cartography/USGS/MapProjections/Azimuthal/Stereographic>

# March-Dollase Function

W. A. Dollase, “Correction of Intensities for Preferred Orientation in Powder Diffractometry: Application of the March Model”, *J. Appl. Cryst.*, **19**(4), 267-272 (1986).

# March-Dollase Function

$$O_{ph} = \frac{1}{M_p} \sum_{j=1}^{M_p} \left( Ratio^2 \cos^2 A_j + \frac{\sin^2 A_j}{Ratio} \right)^{3/2}$$

$h_p$  = reciprocal lattice vector

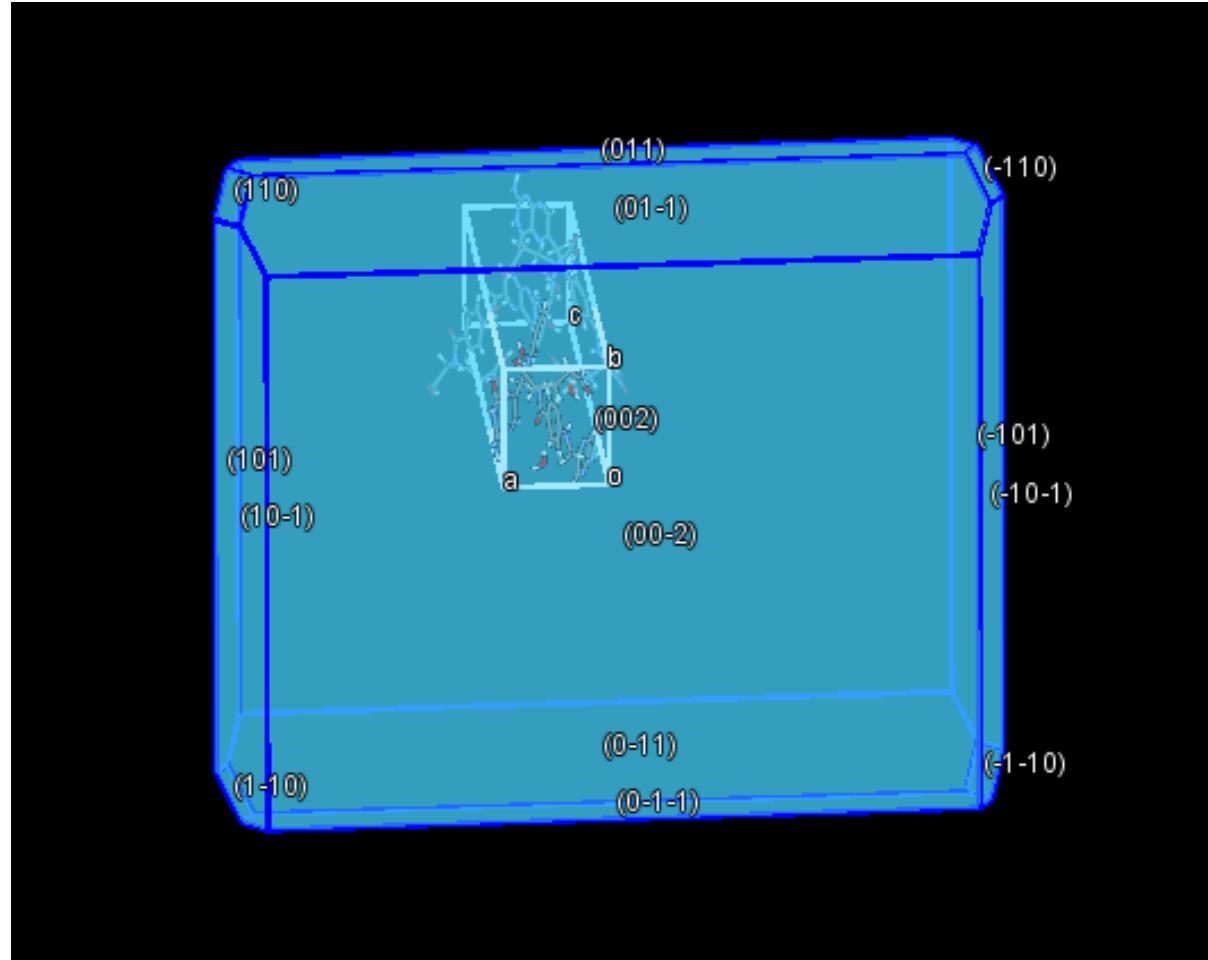
$M_p$  = multiplicity of  $h_p$

$A_j$  = angle between specified unique axis and  $h_p$

Ratio = the refinable parameter “aspect ratio”

Cylindrical specimen symmetry assumed

# BFDH Morphology - Folic Acid Dihydrate

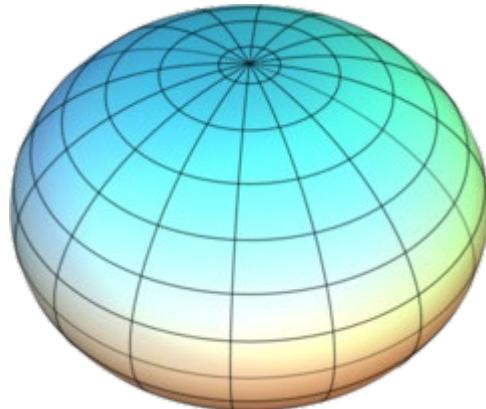


# March-Dollase Function (B-B)

**Plates**

Ratio  $< 1$

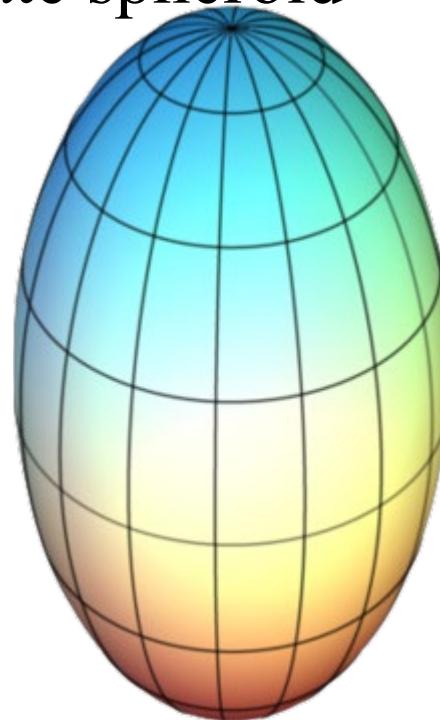
Oblate spheroid



**Needles**

Ratio  $> 1$

Prolate spheroid



Check consistency with anisotropic broadening!

# Spherical Harmonics Function

R. B. Von Dreele, “Quantitative texture analysis by Rietveld refinement”,  
*J. Appl. Cryst.*, **30**, 517-525 (1997).

# Spherical Harmonics Function

$$O_p(h, y) = 1 + \sum_{L=2}^{N_L} \frac{4\pi}{2L+1} \sum_{m=-L}^L \sum_{n=-L}^L C_L^{mn} k_L^m(h) k_L^n(y)$$

Terms depend on crystal and sample symmetry  
cylindrical  
2/m (shear)  
mmm (rolling)  
no symmetry

- Project: C:\zjak01\ICDD\_pharma
  - Notebook
  - Controls
  - Covariance
  - Constraints
  - Restraints
  - Rigid bodies
- PWDR 11bmb\_3158.fxye Bank 1
  - Comments
  - Limits
  - Background
  - Instrument Parameters
  - Sample Parameters
  - Peak List
  - Index Peak List
  - Unit Cells List
  - Reflection Lists
- Phases
  - eltrombopag dioleate

Histogram data for eltrombopag dioleate:  
PWDR 11bmb\_3158.fxye Bank 1

Select plot type:

None  
 Mustain  
 Size  
 Preferred orientation  
 St. proj. Inv. pole figure  
 Eq. area Inv. pole figure

Use Histogram: PWDR 11bmb\_3158.fxye Bank 1 ?  Do new LeBail extraction?

In sequential refinement, fix these in eltrombopag dioleate for this histogram:

Phase fraction: 15.7018 Wt. fraction: 1.000

Domain size model: isotropic  LGmix 1.0

size(μm): 1.0

Mustain model: uniaxial  LGmix 1.0

Unique axis, H K L: 0 1 0

Equatorial mustain: 1262.6  Axial mustain: 1859.1

Hydrostatic/elastic strain:

D11 0.0  D22 0.0  D33 0.0

D13 0.0

Layer displacement (μm): 0.0

Preferred orientation model Spherical harmonics   Refine?

Spherical harmonic coefficients: Texture index: 1.137

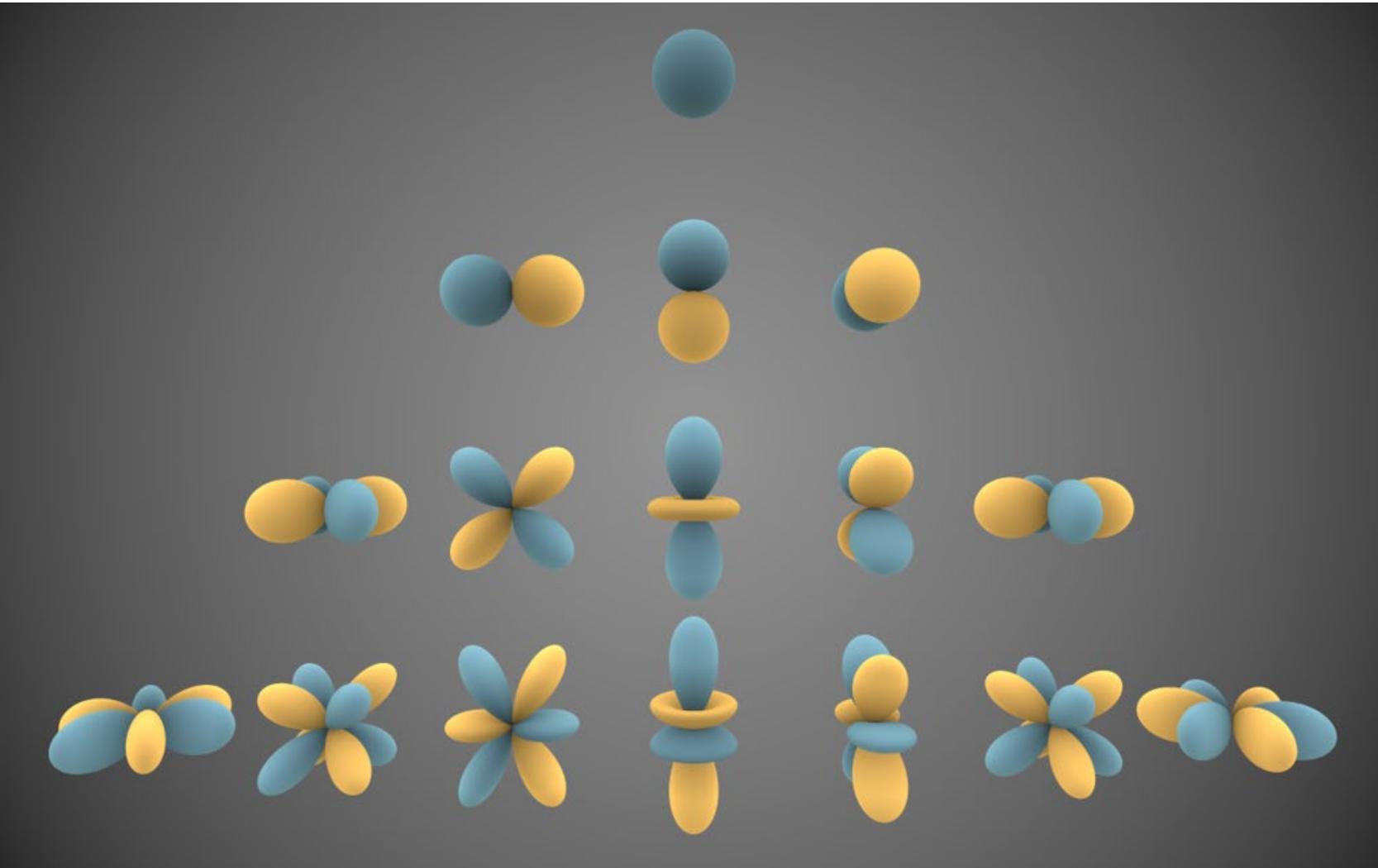
C(2,-2)	-0.479	C(2,0)	0.434	C(2,2)	-0.059	C(4,-2)	-0.513
C(4,-4)	-0.137	C(4,0)	-0.294	C(4,2)	0.163	C(4,4)	-0.277

Negative MRD penalty list:   Zero MRD tolerance: 0.1

Extinction: 0.0

Babinet A: 0.0  Babinet U: 0.0

# Spherical Harmonics



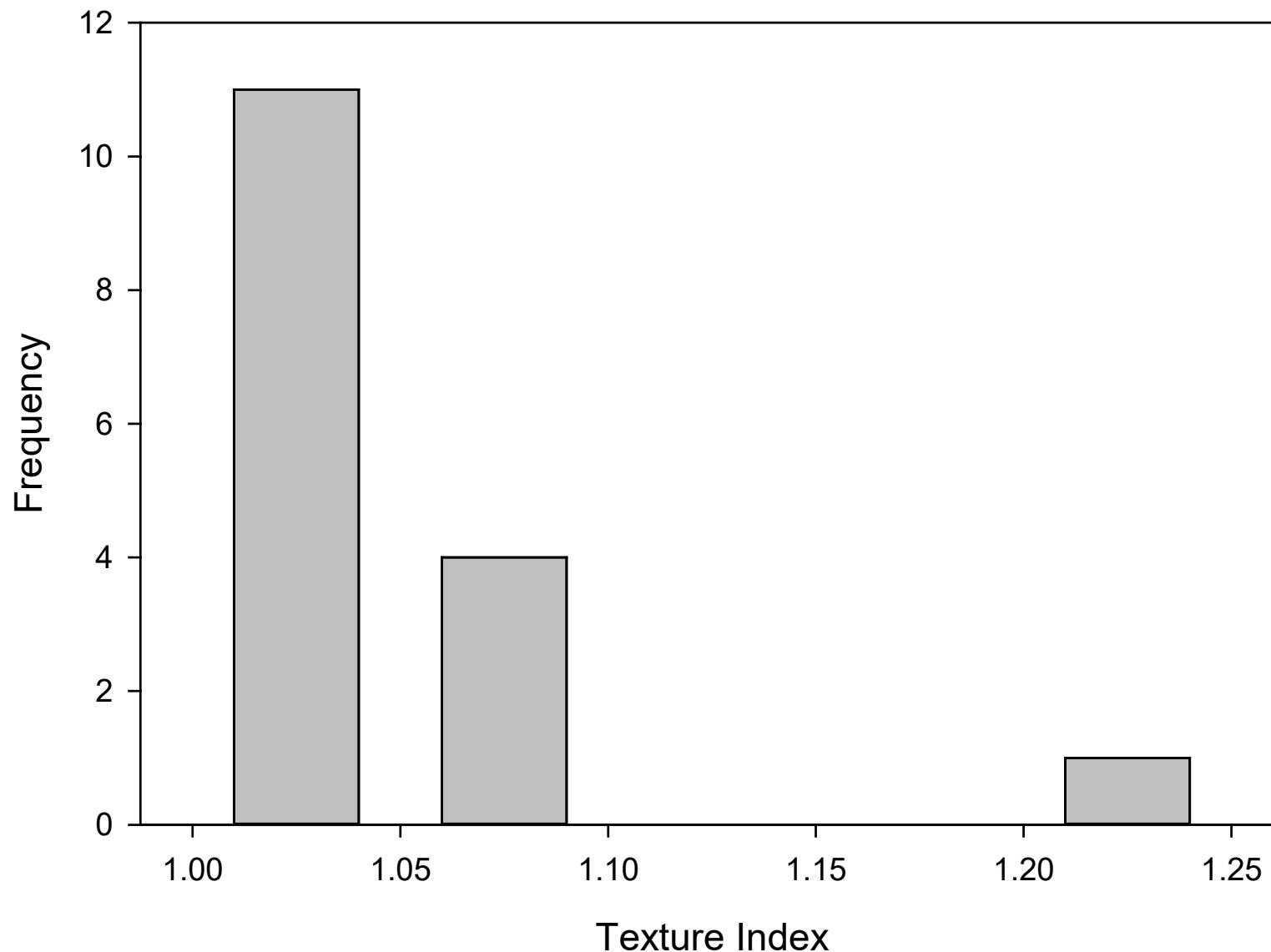
# Texture Index

$$J = 1 + \sum_{L=2}^{N_L} \frac{1}{2L+1} \sum_{m=-L}^L \sum_{n=-L}^L |C_L^{mn}|^2$$

$J = 1$  for random

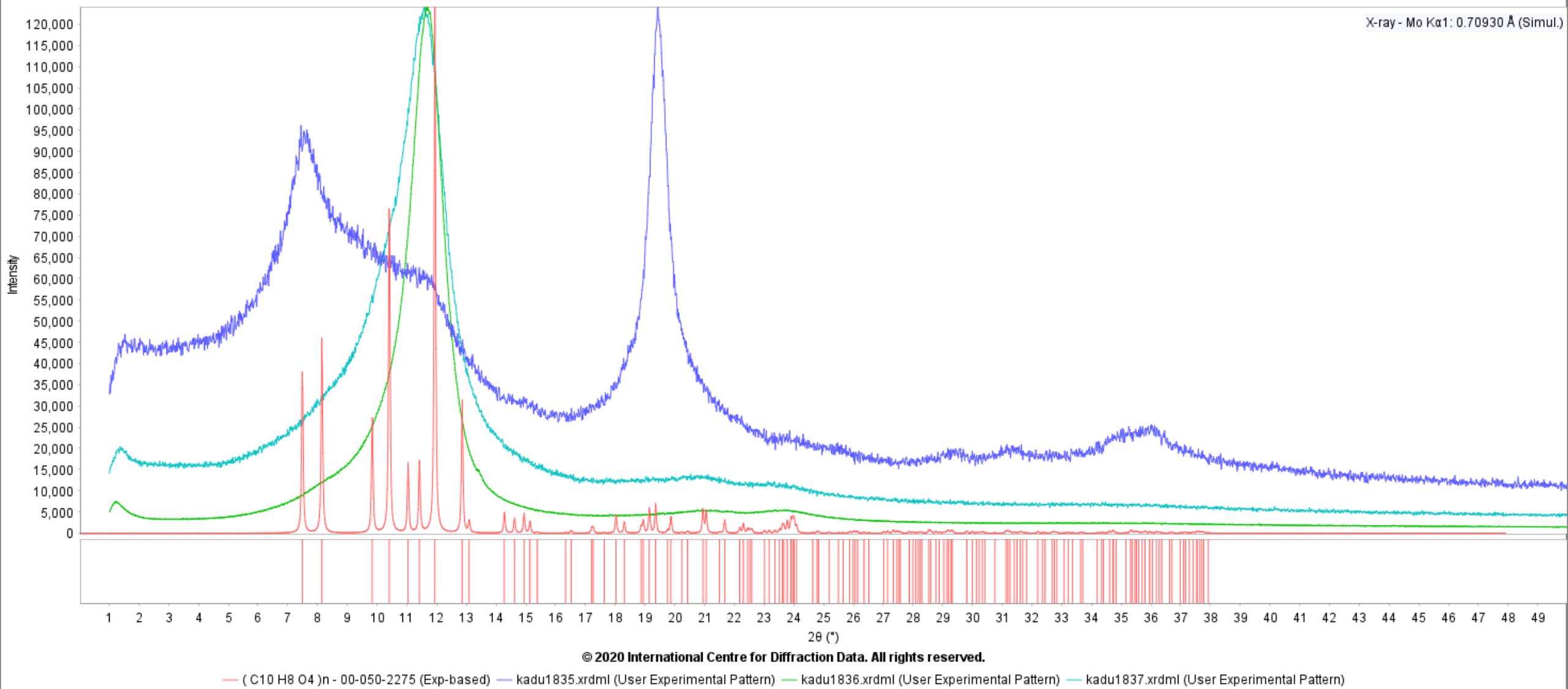
$J = \infty$  for single crystal

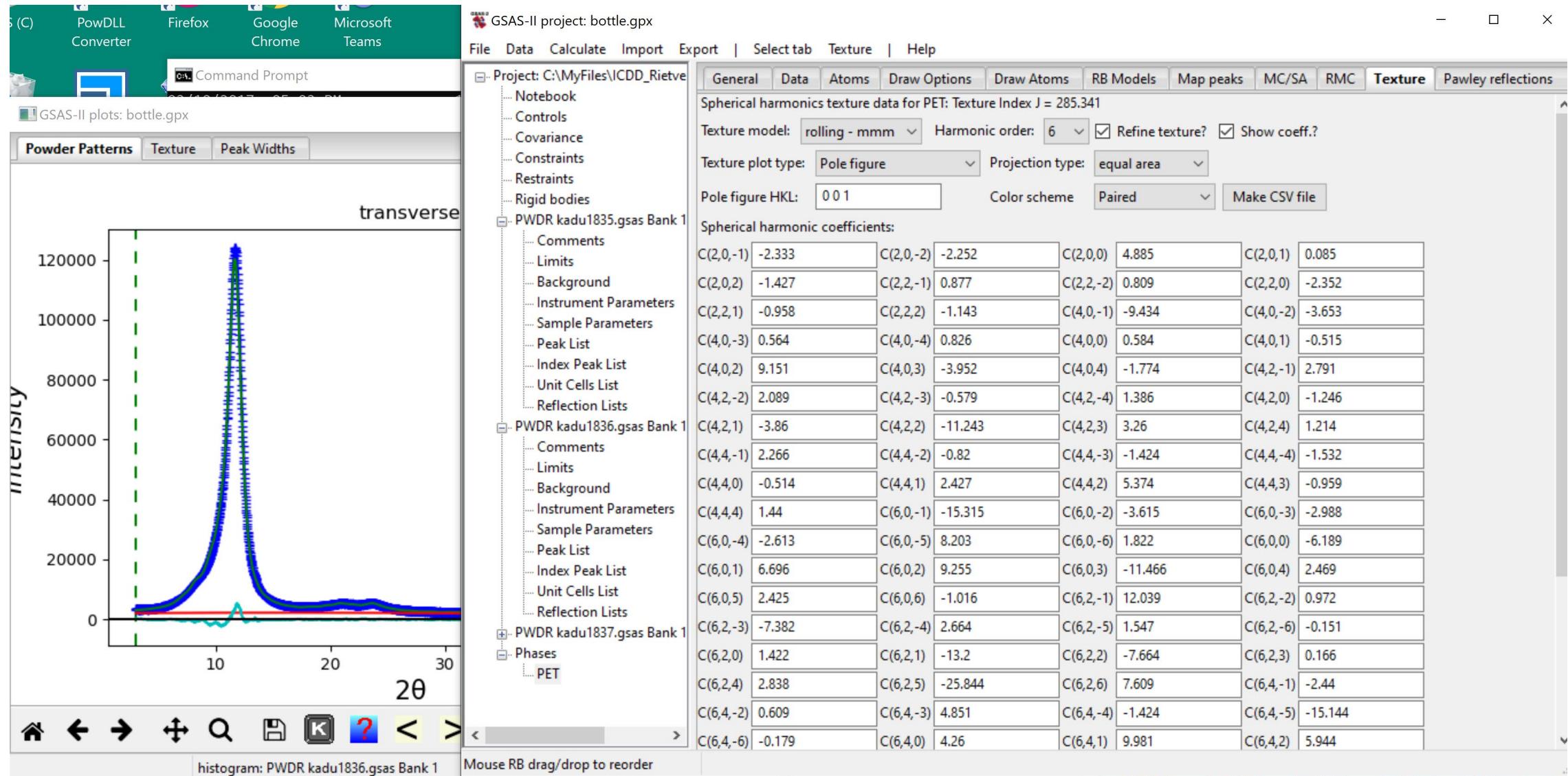
# Texture Index in 11-BM Pharmaceuticals

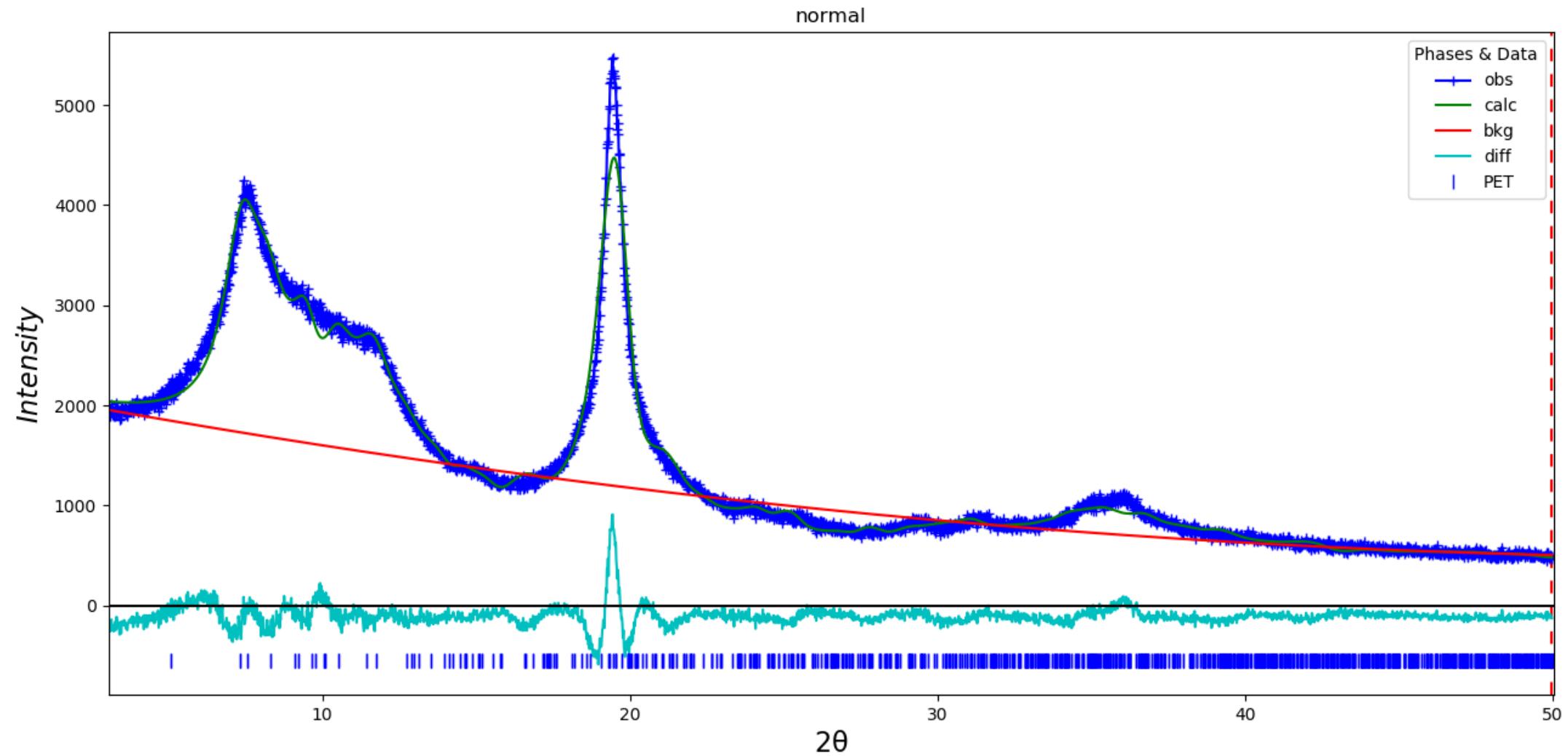


2-L PET Bottle

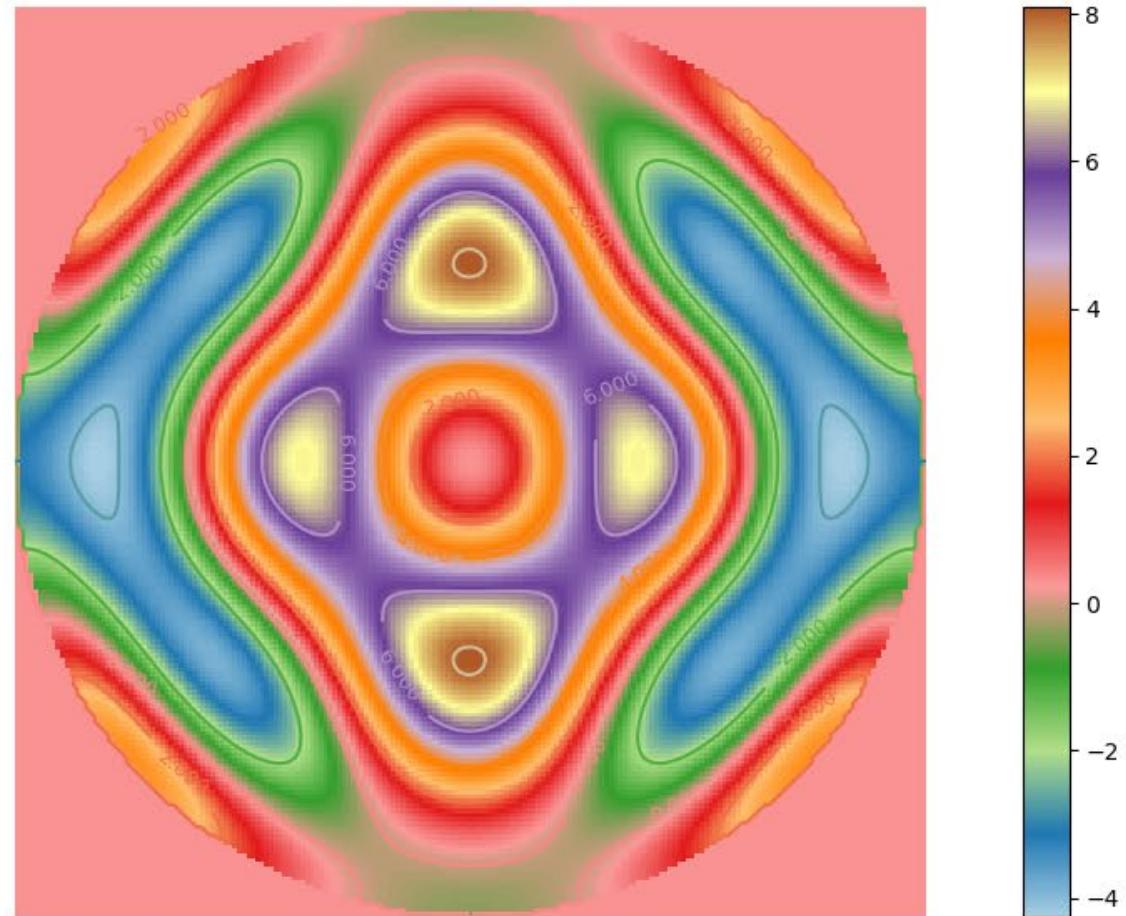
X-ray - Mo K $\alpha$ 1: 0.70930 Å (Simul.)







0 0 1 Pole figure for PET



# Texture in HDPE Pipe

